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Chemical design principles of next-generation antiviral surface coatings

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The ongoing coronavirus disease 2019 (COVID-19) pandemic has accelerated efforts to develop high-performance antiviral surface coatings while highlighting the need to build a strong mechanistic understanding of the chemical design principles that underpin antiviral surface coatings. Herein, we critically summarize the latest efforts to develop antiviral surface coatings that exhibit virus-inactivating functions through disrupting lipid envelopes or protein capsids. Particular attention is focused on how cutting-edge advances in material science are being applied to engineer antiviral surface coatings with tailored molecular-level properties to inhibit membrane-enveloped and non-enveloped viruses. Key topics covered include surfaces functionalized with organic and inorganic compounds and nanoparticles to inhibit viruses, and self-cleaning surfaces that incorporate photocatalysts and triplet photosensitizers. Application examples to stop COVID-19 are also introduced and demonstrate how the integration of chemical design principles and advanced material fabrication strategies are leading to next-generation surface coatings that can help thwart viral pandemics and other infectious disease threats.

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1. Introduction

Various emerging infectious diseases caused by viruses such as severe acute respiratory syndrome-coronavirus (SARS-CoV, also known as SARS-CoV-1), Middle East respiratory syndrome (MERS)-CoV, Ebola virus, influenza virus, norovirus, Zika virus, and dengue virus pose serious threats to human health worldwide.^{1–3} In 2009–2010, a severe flu-like illness caused by a novel H1N1 strain of the influenza A virus spread rapidly



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of membrane-targeting strategies to inhibit membrane-enveloped biological nanoparticles, including viruses.

across more than 214 countries, resulting in the estimated death of over 200 000 people due to the pandemic.⁴ In 2015, the Centers for Disease Control and Prevention (CDC) reported that the fatality rate from an Ebola virus outbreak in West Africa was approximately 70% among diagnosed cases.⁵ Such examples highlight both the broad impact and serious health and biosecurity risks posed by viruses, especially in outbreak situations such as epidemics and pandemics.^{6,7}

The ongoing coronavirus disease 2019 (COVID-19) pandemic is perhaps the greatest virus challenge faced in modern times and is caused by the severe acute respiratory syndromecoronavirus 2 (SARS-CoV-2), which has infected over 174 million people and caused over 3.7 million deaths as of June 2021. While ongoing developments on the vaccine and antiviral therapeutic fronts offer hope for curtailing the pandemic,⁸⁻¹² there continues to be extensive attention placed on stopping community transmission and preventing virus contamination of surfaces.¹³⁻¹⁵ These efforts have led to renewed attention on the surface contamination and decontamination of viruses in general and have also inspired the design of innovative material interfaces to curb transmission of SARS-CoV-2 and other viral threats.¹⁶⁻²³ To date, recent reviews covering various aspects of antiviral surface coatings²⁴⁻³¹ have mainly focused on conceptual descriptions and biological performance results, while there remains an outstanding need to build a deeper mechanistic understanding of how different coatings work in terms of molecular-level chemical principles.

Herein, we critically summarize the recent development of antiviral coating strategies to inactivate viruses on surfaces, including self-cleaning surfaces, along with pertinent coverage of virus stability on surfaces and the latest efforts to inactivate SARS-CoV-2 on surfaces. Particular focus is placed on distilling the chemical design principles that underpin the virus-inactivating properties of antiviral surface coatings and exploring how such mechanistic insights can guide the development of next-generation surface coatings to stop viruses and other infectious pathogens. To create antiviral surface coatings, one may physically disrupt attaching virus particles and/or prevent the attachment process itself, and physicochemical factors such as hydrophobic interactions, electrostatic interactions, hydrogen bonding, and/or redox and coordination reactions between viruses and solid surfaces play important functional roles.^{32,33} Such interactions and/or reactions are influenced by virus structure, material surface chemistry, and environmental conditions.^{34–36}

Before proceeding to discuss virus interactions with surfaces, we briefly discuss the SARS-CoV-2 virus structure to give insight into potential molecular-level targets and broader implications for different types of viruses. SARS-CoV-2 belongs to the genus β-coronavirus and is a single-stranded positivesense (ss-(+)) RNA virus with an approximately 30-kb nucleotide base length and a particle size of 60–100 nm (Fig. 1a).³⁷ SARS-CoV-2 encodes at least 29 proteins, four of which are structural proteins that are needed to produce a structurally complete virus particle and include the following: spike (S) glycoprotein, membrane (M) glycoprotein, envelope (E) glycoprotein, and nucleocapsid (N) phosphoprotein (Fig. 1b). S proteins assemble into homo-trimers composed of three protomers (Fig. 1c). Each S protomer comprises the receptor-binding fragment S1, the fusion fragment S2, and a single transmembrane (TM) anchor. Both S1 and S2 subunits contain a receptor binding domain (RBD), N-terminal domain, and C-terminal domain. The trimeric S protein binds to the host membrane, which is a necessary step for initiating viral fusion.38-40 The RBD has two different conformations, with "up" and "down" representing a receptoraccessible state and a receptor-inaccessible state, respectively.38,41 The N-terminus includes a disulphide bond (Cys15-Cys136) and an N-linked glycan at the Asn17 position (Fig. 1d). The S, M, and E proteins together with the phospholipid (PL) bilayer form the viral envelope, while the N protein interacts with genomic RNA of SARS-CoV-2 and other RNA viruses (or genomic DNA in the case of DNA viruses) to form the viral nucleocapsid.



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Fig. 1 Overview of virus structural components and SARS-CoV-2 example. Representative cryoelectron tomography image of SARS-CoV-2 virus particles (a).³⁷ Schematic illustration of an individual SARS-CoV-2 virus particle, depicting the prefusion spike (S) protein with the receptor-binding domain (RBD) domain in "down" (salmon) and "up" (red) conformations, the phospholipid (PL) bilayer envelope (grey), and ribonucleoproteins (RNPs, yellow) (b).³⁷ Ribbon representation of the S protein structure in the prefusion conformation, including various structural components such as the N-terminal domain (NTD), RBD, C-terminal domain 1 (CTD1), CTD2, fusion peptide (FP), fusion-peptide proximal region (FPPR), heptad repeat 1 (HR1), central helix (CH), and connector domain (CD), together with the N-terminus, S1/S2 cleavage site, and S2' cleavage site (c).⁴⁸ Molecular model of the N-terminal segment of the S protein, where the N-terminus is located at the Gln14 residue after cleavage of the signal peptide and Cys15 forms a disulphide bond with Cys136 (d).⁴⁸ Cross-sectional representation of a SARS-CoV-2 virus particle and relevant chemical structures related to PL molecules, proteins, and nucleic acids are presented (e). While a PL bilayer is found in enveloped viruses only, proteins and nucleic acids are found in both enveloped and non-enveloped viruses.

In general, viruses that have a membrane envelope are classified as enveloped viruses, while those lacking an envelope are referred to as non-enveloped viruses (Fig. 1e). Accordingly, various antiviral surface coatings have been designed for targeting membrane-enveloped and/or non-enveloped viruses. There are three general classes of surface-coating-based mitigation strategies:⁴² (1) disruption of the PL bilayer of the viral envelope; (2) damage of important proteins that are part of the envelope or capsid; and (3) degradation of nucleic acids. The chemical composition and conformation of virus components such as the PL bilayer and proteins influence the physicochemical properties and stability of viruses.⁴³ For example, the PL bilayer imparts negative surface charge and hydrophobic properties to virus particles. The conformation of viral proteins is also critical for necessary functions in the virus life cycle and disrupting key structural elements in viral proteins can therefore abrogate infectivity along with impairing viral nucleic acids involved in viral genome replication.^{44,45} In the latter respect, the presence of reactive species such as Ag^+ , hydroxyl radical (*OH), and singlet oxygen (¹O₂) strongly influences virus inactivation efficiency^{46,47} and is an important design consideration for antiviral surface coatings.

2. Virus attachment and stability on surfaces

Viruses cannot replicate independently without a host cell, and can survive on inanimate surfaces for <5 min to >28 days, depending on surface properties, environmental conditions such as temperature, humidity, and light, and the type and amount of virus contaminant (Table 1).^{42,49–56} Typically, enveloped viruses are less stable than non-enveloped viruses, because the PL bilayer has high susceptibility to physical disruption and the

Virus	Type ^{<i>a</i>}	Diameter (nm)	Survival time [d (days), h, or min] (temperature and relative humidity (RH)) on surfaces	Ref.
SARS-CoV-2	ssRNA, E	60-140	3 d (21-23 °C, 40%) on plastic PP; 2 d (21-23 °C, 40%) on stainless steel; 4 h (21-23 °C, 40%) on Cu:	49
			 a h (22 °C, 65%) on printing and tissue paper; b (22 °C, 65%) on wood and cloth; c (22 °C, 65%) on glass and banknotes; c (22 °C, 65%) on stainless steel and plastic; c (22 °C, 65%) on the outer layer of a surgical mask; 	75
			> 2 d (24 °C, 20%) on stainless steel; > 1 d (24 °C, 40%) on stainless steel; > 1 d (24 °C, 60%) on stainless steel; > 0.3 d (35 °C, 60%) on stainless steel;	76
			 >28 d (20 °C, 50%) on stainless steel; >7 d (30 °C, 50%) on stainless steel; >1 d (40 °C, 50%) on stainless steel; 	77
			0.5 h (65 °C, 94%) on N95 respirators	78
SARS-CoV-1	ssRNA, E	80-160	5 d (22–25 °C, 40–50%) on plastic	70
SARS-CoV-1	ssRNA, E	80-160	1 d (20 °C) on paper and cotton gown; >2 d (20 °C) on disposable gown	53
Human CoV-229E	ssRNA, E	80-160	5 d (21 °C, 30–40%) on PTFE, PVC, ceramic tiles, glass, and stainless steel; 3 d (21 °C, 30–40%) on silicon rubber; 2 h, 40 min, and 30 min (21 °C, 30–40%) on Cu–Ni alloy with 70%, 79%, and 80% Cu; 5 min (21 °C, 30–40%) on cartridge brass with 70% Cu and 30% Zn; <5 min (21 °C, 30–40%) on Cu	42
MERS-CoV	ssRNA, E	120-160	Half-life of 0.85 h (20 °C, 40%), 0.44 h (30 °C, 30%), and 0.90 (30 °C, 80%) on plastic, Half-life of 0.94 h (20 °C, 40%), 0.97 h (30 °C, 30%), and 0.64 (30 °C, 80%) on staipless steel	79
Respiratory syncytial virus	ssRNA, E	120-300	>1 d $(r.t.)^{b}$ on Al alloy $(R_{rms}^{c} = 0.6 \pm 0.1 \text{ nm});$ 1 d $(r.t.)$ on etched Al alloy $(R_{rms} = 995 \pm 115 \text{ nm})$	64
Ebola virus	ssRNA, E	60-80	3d (27 °C, 80%) and 3 d (21 °C, 40%) on utility-grade (308) stainless steel; 3 d (27 °C, 80%) and 11 d (21 °C, 40%) on PTFE	50
Influenza A virus (H1N1)	ssRNA, E	80-120	2 d (r.t.) on wooden chopping board; 1 d (r.t.) on stainless steel and plastic; 8 h (r.t.) on cotton pillowcase	80
Influenza A virus (H1N1)	ssRNA, E	80-120	>1 d (27 °C, 50–60%) on stainless steel; 6 h (27 °C, 50–60%) on Cu	52
Parainfluenza virus	ssRNA, E	120-300	10 h (r.t.) on stainless steel and laminated plastic; 6 h (r.t.) on hospital gown, facial tissue, and laboratory coat	81
Hepatitis A virus	ssRNA, NE	20-50	9.0 d (15 °C, 30%) on chopping board; 5.4 d (15 °C, 30%), 8.4 d (15 °C, 70%), and 0.06 d (45 °C, 30%) on stainless steel	54
Hepatitis A virus	ssRNA, NE	20-50	Half-life of 169 h (5 °C, 25%), 151 d (5 °C, 50%), and 50 h (20 °C, 55%) on stainless steel	71
Mouse hepatitis virus	ssRNA, E	60-160	28 d (4 °C, 20–80%) and 5 d (20 °C, 50%) on stainless steel	51
Murine norovirus	SSRNA, NE	28-35	4.8 d (15 °C, 30%) and 6.6 d (15 °C, 70%) on chopping board	54
Rovine rotavirus	SSKINA, NE	28-35 70	14 u (22 \odot) on stainless steel 14 d (22 $^{\circ}$ C) on stainless steel	82 82
Bacteriophage MS2	ssRNA, NE	24–27	8.4 d (15 °C, 30%) and 11.4 d (15 °C, 70%) on chopping board; 19.8 d (15 °C, 30%) and 9.0 d (15 °C, 70%) on stainless steel	54

Table 1 Representative examples of virus survival on surfaces in different environmental conditions

^{*a*} ssRNA, dsRNA, E, and NE represent single-stranded RNA, double-stranded RNA, enveloped virus, and non-enveloped virus, respectively. ^{*b*} Room temperature. ^{*c*} Root-mean-square roughness.

functional conformation of envelope proteins depends on an intact bilayer environment.⁵⁷ Since enveloped viruses are surrounded by a PL bilayer and several structural proteins, they possess multiple patches of hydrophobic, hydrophilic, and positive and/or negative charges due to the presence of the PL headgroup, alkyl chains, and NH_3^+ (or NH_2) and COO^- (or COOH) groups.³² Viruses adsorb onto a solid surface through electrostatic, hydrophobic, and/or van der Waals interactions, according to the extended Derjaguin–Landau–Verwey–Overbeek (EDLVO) theory.^{58,59} For solid surfaces having hydroxyl functional groups such as glass, paper, wood, and cotton, an additional interaction, hydrogen bonding, should also be accounted, while steric effects and surface roughness should be considered for porous fomite surfaces such as paper, cardboard, and ${\rm cloth.}^{60,61}$

In general, viruses are more stable on smooth surfaces than on porous surfaces.^{62,63} As the root-mean-square (RMS) roughness of an alumina alloy surface increased from 0.6 to 995 nm, the mean tissue culture infective dose (TCID₅₀) of respiratory syncytial virus on the surfaces decreased from 10^3 to 0 TCID₅₀ mL⁻¹, which occurred due to entrapment and disruption of virus on nanostructured hydrophobic surfaces.⁶⁴ SARS-CoV-1 and SARS-CoV-2 survived for 3 and 2 days on surfaces of polypropylene (PP) plastic and stainless steel, respectively, while no viable virus was observed on a Cu surface after 4 h.⁴⁹ The survival time of human coronavirus 229E on Cu alloys was shorter than on stainless steel depending on the ratio of Cu,⁴² due to the leaching of antiviral cuprous and cupric ions (Cu⁺ and Cu²⁺, respectively) that were released from Cu.^{42,65} The viral titre of influenza A virus (H1N1) at 22 °C and with humidity of 50–60% decreased from 2×10^6 to 500 and 5×10^5 plaque-forming units (PFU) mL⁻¹ after 24 h on Cu and stainless steel surfaces, respectively.⁵² The viral titre of influenza A virus (H1N1) decreased from 2×10^6 to 500 and 5×10^5 PFU mL⁻¹ after 24 h on Cu and stainless steel surfaces, respectively.⁵² The viral titre of conduct time of human coronavirus 229E on Cu alloys was shorter than on stainless steel depending on the ratio of Cu,⁴² due to the leaching of antiviral cuprous and cupric ions (Cu⁺ and Cu²⁺, respectively) that were released from Cu.^{42,65}

Temperature and humidity both influence virus survival.^{51,66,67} Viruses containing protein, RNA, or DNA are typically disrupted at >50 °C, and heat treatment at >60 °C for >1 h is sufficient to inactivate most viruses.⁶⁸ Treatment at 56 °C for 45 min completely inactivates SARS-CoV-2 with an initial viral titre of 10⁶ PFU mL⁻¹.⁵⁶ As the temperature decreases from 34 °C to 14 °C at a relative humidity (RH) of 75%, the viral infectivity of the Phi6 bacteriophage (another type of virus that infects bacteria) increases exponentially by more than six orders of magnitude.⁶⁹ The influence of humidity on virus survival is neither monotonic nor consistent, while both enveloped and non-enveloped viruses with lipid envelopes and protein capsids, respectively, survive for shorter periods at moderate humidity (40–70%) than at low ($\leq 20\%$) and high (\geq 80%) humidity.^{51,54,68,70} Virus inactivation on surfaces involves desiccation-induced oxidation and Maillard reactions^{71,72} as well as structural damage of lipid envelopes and protein capsids at the air-water interface in a humiditydependent manner.⁷³ Temperature and humidity both influence virus survival, which also depends on the surface properties of the materials onto which a virus attaches.^{51,54,68-70} SARS-CoV-1 on the dried surfaces of plastic plates survives for >5 days in a typical air-conditioned environment at 22–25 $^\circ\!\mathrm{C}$ and with humidity of 40–50%, while there was a viral titre reduction of $10^{3.38}$ -fold over 2 days at 38 °C and >95% humidity.⁷⁰ Under simulated sunlight conditions, 90% of SARS-CoV-2 virus in simulated saliva dried on grade 304 stainless steel coupons was inactivated after 6.8 and 12.8 min UVB (280-315 nm) light irradiation at 0.16 and 0.03 mW cm^{-2} , respectively, while no significant decrease in viral titre ($10^{3.0\pm0.2}$ TCID₅₀ mL⁻¹) occurred on the surface after 30 min drying compared with that before drying $(10^{2.8\pm0.1} \text{ TCID}_{50} \text{ mL}^{-1})$ in the dark.⁷⁴ Such findings are not only important for understanding virus-material interactions but can also guide material design to stop virus spreading in real-life settings.

Respiratory and enteric viruses spread directly between people and can also spread indirectly through virus-contaminated surfaces. In high-risk environments such as hospitals and public facilities, virus contaminants on high-touch surfaces such as wards, handrails, switches of lights and fans, lift buttons, and toilets spread to the hands of healthcare workers, patients, and visitors (Fig. 2).⁸³ Personal protective equipment (PPE) such as facial masks, protective suits, and gloves helps healthcare workers to avoid infection. However, PPE still imposes a risk of virus



Fig. 2 Virus contamination on different types of surfaces. The extent of time during which a virus survives, *i.e.*, remains infective, depends on surface properties, environmental conditions such as temperature and humidity, and virus type.

transmission, especially during the PPE removal step, because respiratory droplets can settle on the surface of used PPE.^{84,85} Routine disinfection and cleaning of high-touch surfaces with UV light irradiation and chemicals are important and necessary for preventing virus transmission. A 9 min exposure to UV light at 254 nm and 1.94 mW m⁻² completely inactivated a 600 μ L SARS-CoV-2 solution with a concentration of 5×10^{6} TCID₅₀ mL⁻¹ in a 24-well plastic plate, which demonstrated greater efficiency than the 10-fold reduction observed under UVA light irradiation at 365 nm and 0.54 mW m^{-2.86} A 20 s exposure to UV light at 250–300 nm and 3.5 mW m⁻² reduced the viral titre of 150 μ L SARS-CoV-2 in a 60 mm plastic Petri dish by >10^{3.3}-fold.⁸⁷ Although the UV treatment of N95 masks and respirators can reduce the viral load of SARS-CoV-2 by >104-fold on N95 respirators and glass slides within 5 min,⁸⁸ viral RNA is retained.⁸⁹ Most commonly used self-cleaning materials are oxidants such as hypochlorous acid and hydrogen peroxide as well as alcohols such as ethanol and propanol.⁹⁰ Disinfection with materials such as 70% ethanol or 0.5% povidone-iodine for 15 s has also reduced the viral titre of SARS-CoV-2 in aqueous solution by $> 10^{4.33}$ -fold, ⁹¹ while a 5 min contact time with 10% bleach (0.6% hypochlorite), 3% H₂O₂, and a commercial QAC, Sani-Cloth AF3 germicidal disposable wipe, reduced the titre of SARS-CoV-2 on 3D-printed surgical masks by $> 10^5$ -fold.⁹² Since disinfectant processes are labour-intensive for adequate cleaning due to chemical issues such as dewetting and volatility, self-cleaning surfaces are also being explored as a promising alternative method to prevent virus contamination by repelling virus attachment and inactivating attached virus particles.

Repelling of viruses on superhydrophobic (SHPB) surfaces is inspired by self-cleaning phenomena found in nature such as on the surfaces of lotus leaves where water droplets roll off and take away dust without leaving any trace contaminants.⁹³ SHPB surfaces with water contact angles (θ_{WCA}) > 150° and rolling angles (θ_{RA}) < 10° (Fig. 3a) are typically designed by decreasing surface energy and increasing surface roughness to produce an "air cushion" between the water layer and surface in the Cassie– Baxter wetting regime (Fig. 3b),^{94–96} avoiding virus attachment on SHPB surfaces.⁹⁷ Polytetrafluoroethylene (PTFE) nanoparticles (NPs) with 250 nm diameter were sintered into PP textiles made from 2–8 µm fibres to produce PTFE/PP textiles with a doubleroughened structure and low-surface energy functional groups of CF₂, CF₃, and CF₃CF₂ with 18, 6, and 17 mN m⁻¹, respectively,⁹⁸



Fig. 3 Surface decontamination of viruses by repelling virus contaminants on superhydrophobic (SHPB) surfaces with water contact angle (θ_{WCA}) > 150° and rolling angle (θ_{RA}) < 10° (a). Repulsion of water droplet by air cushion between water and SHPB surface in the Cassie–Baxter wetting regime (b).

and $\theta_{\rm WCA}$ = 160.8 \pm 2.3 $^{\circ}$ and $\theta_{\rm RA}$ = 2.8 \pm 1.4 $^{\circ}$ indicated formation of an SHPB surface.⁹⁴ The PTFE/PP textiles prevented the attachment of adenovirus type 4 and 7a virus particles by $10^{2.1}$ - and $10^{1.6}$ -fold, respectively, compared to non-coated PP textiles. A nanostructured glass surface with nanopillars of 430 nm height and 100 nm diameter was prepared by a metal dewetting method and coated with trichloro(dodecyl)silane (SiCl₃(CH₂)₁₁CH₃, TCDDS) to produce an SHPB surface with oleophilicity ($\theta_{WCA} = 154^{\circ}$ and $\theta_{CA} = 26^{\circ}$ in hexadecane), and reduced viral infectivity to <10% (= inactivating activity of >90%) for influenza A virus (H3N2), as compared to <25-78% for other hydrophilic/oleophobic and hydrophobic/oleophobic surfaces with θ_{WCA} = 78–140° and θ_{CA} = 20–95° in hexadecane.99 The difference in antiviral performance was attributed to the oleophilic surface possessing a higher affinity to interact with the fatty acid tails of PL than oleophobic surfaces, which resulted in greater disruption of the lipid membrane of enveloped viruses. It should be noted that SHPB surfaces do not inactivate virus, leading to no loss in the transferable amount of virus and no decrease in the risk of viral infection. Thus, more efforts are being directed at developing antiviral surfaces that can inactivate attached virus particles to mitigate infection risk.

3. Chemically functionalized surfaces for virus inactivation

Since viruses survive on surfaces depending on the surface properties (*cf.* Table 1), there has been extensive interest in developing functional surfaces that are coated with antiviral materials and can inactivate viruses to curb viral transmission. Antiviral material options are categorized as organic and inorganic compounds as well as metals such as quaternary ammonium compounds (QACs),¹⁰⁰ porphyrins,¹⁰¹ TiO₂,¹⁰² Ag,¹⁰³ and Cu^{104,105} that act through different antiviral mechanisms depending on their molecular properties. Since the infectivity of enveloped and non-enveloped viruses depends on the presence of intact lipid envelopes and protein capsids, respectively, the interaction of these viral components with antiviral materials on surfaces is required for virus inactivation. The most popular organic antiviral materials are QACs that have positively charged quaternary ammonium groups and hydrophobic alkyl chains on

the ammonium N atom such as alkyl or heterocyclic groups that form a lipophilic tail, which interacts with enveloped viruses through electrostatic and hydrophobic interactions that cause envelope disruption and virus particle inactivation.¹⁰⁶⁻¹⁰⁸ Metal ions such as Ag⁺ and Cu²⁺ released from metallic compounds and metal NPs, coated on surfaces, inactivate enveloped viruses through reactions with the thiol and disulfide bonds (SH and S-S, respectively) of proteins and enzymes¹⁰⁹⁻¹¹² on lipid envelopes, and through the degradation of the helical structure of viral nucleic acids.^{113,114} Practically, Cu-coated surfaces have been reported to reduce the transmission of hand-transmitted health care-associated infections by approximately 44-fold compared to control tests on stainless steel surfaces during past outbreaks caused by influenza A virus, adenovirus, and norovirus.¹¹⁵ AgNPs and Ag compounds such as Ag₂O inactivate viruses due to released Ag⁺.^{105,116,117} Cu and cuprous compounds such as Cu₂O also inactivate viruses due to released Cu⁺ and Cu²⁺,^{42,118} while cupric compounds such as CuO have no or less antiviral activity. 105,116,118,119 The antiviral activity of Cu⁺ is explained by the oxidative disruption properties of •OH and superoxide $(O_2^{\bullet-})$ with oxidation potentials of $E_{ox}(^{\bullet}OH/H_2O) = 2.2-$ 2.6 V and $-0.28 \text{ V} (O_2/O_2^{\bullet-}) \nu s$. normal hydrogen electrode (NHE), respectively,^{120,121} which are generated through the redox reaction of water and O_2 with $Cu^{+52,104,105,119}$ as well as through the disruption of proteins upon reaction of Cu⁺ with S atoms of SH and amino groups (NH₂) of cysteine amino acid residues.^{118,122}

3.1. QAC-coated surfaces

QACs such as didodecyldimethylammonium bromide (DDAB) and *N*-alkylated polyethylenimines (PEIs)^{100,123,124} inactivate enveloped viruses such as influenza A virus and pseudorabies virus,^{108,124–126} but are less active against non-enveloped viruses such as feline calicivirus, poliovirus, and human adenovirus (Table 2).^{124,127} Enveloped viruses have a hydrophobic lipid envelope membrane that is mainly composed of a negatively charged PL bilayer and several types of structural proteins (such as S, M, and E proteins for coronaviruses), while non-enveloped viruses have hydrophilic protein capsids composed of amino acids.^{106,124}

For enveloped viruses, QAC-coated surfaces interact with the PL bilayer envelope through electrostatic interactions with negatively

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Antiviral material	Virus ^{<i>a</i>} $(N_0 [PFU mL^{-1}])^b$	Virus reduction ratio, ^{<i>c</i>} $log_{10}(N_0/N_t)$ (exposure time)	Ref.
Didodecyldimethylammonium bromide (DDAB)-adsorbed SNP-coated glass	IAV (H1N1) (10^6)	>99%, >4.5 (6 h)	123
QAC-containing acrylic melamine-coated film	IAV (H1N1) ($>10^4$)	>99%, >2 (20 min)	133
$C_8H_{17}(CH_3)_2N^+(CH_2)_2OCH_3$ -containing latex paint	Pseudorabies virus (1.0×10^7)	>99.9%, 4.0 (2 h)	127
$\mathrm{C_{16}H_{33}(CH_3)_2N^{+}(CH_2)_2OCH_3\text{-}containing \ latex \ paint}$	Feline calicivirus (1.0×10^7) Pseudorabies virus (1.0×10^7) Feline calicivirus (1.0×10^7)	0% (2 h) >99.9%, 7 (2 h) 0% (2 h)	127
$\rm C_{16}H_{33}(CH_3)_2N^+(CH_2)_2O~(CH_2)_2OCH_3\mbox{-}containing latex paint$	Pseudorabies virus (1.0×10^7) Feline calicivirus (1.0×10^7)	>99.9%, 6.9 (2 h) 0% (2 h)	127
QAC-benzophenone-based ester (BEst)-coated polyurethane (PU)	IAV (H1N1) (4.4×10^4)	99.9%, 3 (0.5 h)	135
QAC-benzophenone-based amide (BAm)-coated PU	IAV (H1N1) (4.4×10^4)	>99.9%, 4 (0.5 h)	135
<i>N</i> , <i>N</i> -Dodecylmethyl polyethylenimine (PEI)-coated glass	IAV $(H1N1)$ (2×10^5)	>99.9%, >4 (0.5 h)	100
N-(15-Carboxypentadecyl) PEI-coated glass	IAV (H1N1) (2×10^5)	>99.9%, >4 (0.5 h)	100
N-(11-Carboxyundecanoyl) PEI-coated glass	IAV $(H1N1)$ (2×10^5)	66%, 0.47 (0.5 h)	100
N-(Undecanoyl) PEI-coated glass	IAV $(H1N1)$ (2×10^5)	6%, 0.025 (0.5 h)	100
Hyperbranched QAC polymer-coated glass	IAV $(H1N1)$ $(4 \times 10^6)^d$	>99.9%, 5.0 (1 h)	124
	Poliovirus Sabin1 $(1.6 \times 10^7)^d$	68%, 0.5 (1 h)	
Hyperbranched OACs polymer-coated plastic	IAV (H1N1) $(4 \times 10^6)^d$	>99.9%, 3.5 (1 h)	124
	Poliovirus Sabin1 $(1.6 \times 10^7)^d$	37%, 0.2 (1 h)	
N.N-Dodecylmethyl PEI-coated PE	Poliovirus (2.2×10^4)	>99.9%, 4 (0.5 h)	125
, , ,	Rotavirus (5.1×10^3)	>99.9%, 3(0.5 h)	
<i>N</i> , <i>N</i> -Hexylmethyl-PEI-coated glass	Poliovirus (1×10^3)	>99.9%, 3(0.5 h)	125
N.N-Dodecylmethyl-PEI-coated glass	IAV (H3N2) (1.2×10^6)	>99.9%, 3.2 (0.5 h)	126
,	IAV (H1N1) (6.6×10^6)	>99.9%, 4.6 (0.5 h)	
N.N-Dodecylmethyl PEI-coated glass	IAV (H3N2) (4.8×10^5)	>99.9%, 3.5 (5 min)	140
, , , , , , , , , , , , , , , , , , , ,	IAV $(H4N2)$ (6.1×10^6)	>99.9%, 4.5 (5 min)	
N.N-Dodecylmethyl PEI-coated glass	IAV $(H1N1)$ (2.0×10^5)	>99.9%, 4.3 (5 min)	107
N.N-Dodecylmethyl PEI-coated PP	IAV $(H1N1)$ (2.0×10^5)	>99.9%, 4.3 (5 min)	107
N.N-Dodecylmethyl PEI-coated PE	IAV $(H1N1)$ (2.0×10^5)	>99.9%, 4.3 (5 min)	107
Permethyl PEI-coated glass	(H1N1) (10 ⁸)	92.1%, 1.10 (10 min)	108
	(H3N2) (10 ⁸)	96.5%, 1.45 (10 min)	
<i>N</i> -Hexyl PEI-coated glass	$IAV(H1N1)(10^8)$	99.8%, 2.65 (10 min)	108
	$IAV(H3N2)(10^8)$	99.3%, 2.15 (10 min)	
N-Dodecyl PEI-coated glass	$IAV(H1N1)(10^8)$	99.4%, 2.20 (10 min)	108
8	$IAV(H3N2)(10^8)$	99.8%, 2.80 (10 min)	
N-Hexadecyl PEI-coated glass	$IAV(H1N1)(10^8)$	49.9%, 0.30 (10 min)	108
	(H3N2) (10 ⁸)	84.2%, 0.80 (10 min)	
N.N-Hexylmethyl PEI-coated glass	$IAV(H1N1)(10^8)$	>99.9%, 3.42 (10 min)	108
, , , , , , , , , , , , , , , , , , , ,	(H3N2) (10 ⁸)	> 99.9% 3.75 (10 min)	
N.N-Dodecylmethyl PEI-coated glass	$IAV(H1N1)(10^8)$	>99.9%, 3.45 (10 min)	108
, · · · · · · · · · · · · · · · · · · ·	$IAV(H3N2)(10^8)$	>99.9%, 4.20 (10 min)	
N.N-Hexadecylmethyl PEI-coated glass	$IAV(H1N1)(10^8)$	49.9%, 0.30 (10 min)	108
·,····································	$IAV(H3N2)(10^8)$	88.0%, 0.92 (10 min)	
N.N-Hexanovlmethyl PEI-coated glass	IAV $(H1N1)$ (10^8)	99.8%, 2.60 (10 min)	108
,	$IAV (H3N2) (10^8)$	99.6%, 2.45 (10 min)	200
N.N-Dodecanovlmethyl PEI-coated glass	$IAV (H1N1) (10^8)$	99.8%, 2.75 (10 min)	108
	$IAV (H3N2) (10^8)$	99.6%, 2.45 (10 min)	100
PEI (5 layers)-coated polyethersulfone (PES) membranes	MS2 bacteriophage (4.0×10^8)	99%, 2 (0.5 h)	143

^{*a*} IAV represents influenza A virus. ^{*b*} N_0 is the number of virus particles [PFU mL⁻¹] before exposure unless specified otherwise. ^{*c*} Reduction ratios are calculated from $(1 - N_t/N_0) \times 100\%$ and $\log_{10}N_0 - \log_{10}N_t (= \log_{10}(N_0/N_t))$, where N_t is the number of infectious virus particles after exposure to an antiviral surface for time *t*. ^{*d*} N_0 is the number of virus particles with the unit of TCID₅₀ mL⁻¹ before exposure.

charged PL headgroups and hydrophobic interactions with PL alkyl chains, which cause envelope disruption and virus inactivation (Fig. 4a).¹⁰⁷ The PL bilayer contains two parallel layers of PL molecules with the hydrophilic headgroups directed outwards and the hydrophobic fatty acid tails directed towards one another and forming the inner core (Fig. 4b).¹²⁸ The bilayer thickness is typically in the range of 3.6–4.0 nm for most enveloped viruses.¹²⁸ Despite this common structural feature, it is noted that the inactivation efficiency is influenced by the type of enveloped virus, which highlights the importance of additional structural elements such as the specific PL composition and envelope protein density^{129–131} (Table 2). Therefore, empirical testing is warranted to identify the most effective QAC to inhibit a

certain type of virus (Fig. 4c and d). By contrast, for non-enveloped viruses, the cationic quaternary ammonium groups of QACs can only interact with hydrophilic protein capsids, resulting in a lower degree of antiviral activity since the virus particles are not fully disrupted.^{124,127}

For practical use in conventional paints such as anionic acrylates (latex paints), commercial QACs such as benzalkonium chloride and didecyldimethylammonium chloride need to be mixed homogeneously with paints or monomers of latex films.^{127,132,133} To improve compatibility, amphiphilic QACs with alkyl and methoxyl groups, $R(CH_3)_2N^+(CH_2)_2OX$, $R = C_8H_{17}$ or $C_{16}H_{33}$, $X = CH_3$ or $(CH_2)_2OCH_3$, have been used to prepare QAC-coated paints (Fig. 5a), and can reduce the viral titre by 10^4-10^7 -fold for



Fig. 4 Inactivation of enveloped viruses by disrupting the PL bilayer envelope through interactions with quaternary ammonium compounds (QACs) coated on the surface (a and b). Scanning electron microscopy (SEM) images of influenza virus particles after exposure to bare (c) and *N*,*N*-dodecyl, methyl-PEI-coated (d) silicon wafers.¹⁰⁷ The latter case is a representative QAC-coated surface that caused gross morphological disruption of virus particles.



Fig. 5 Examples of several QACs and composites that are practically useful for virus inactivation on QAC-coated surfaces.

enveloped viruses such as pseudorabies virus, depending on the R substituent group of the QACs.¹²⁷ C₁₆-QACs with longer C₁₆ chains have higher hydrophobicity (partition coefficient, log $K_{ow} = 8.25$ and 4.00 for *n*-hexadecane and *n*-octane, respectively)¹³⁴ and antiviral activity (10⁷-fold reduction in viral titre) than C₈-QACs with shorter C₈ chains (10⁴-fold reduction in viral titre). On the other hand, QAC-coated paints have no or less activity against non-enveloped viruses such as feline calicivirus. In terms of applications, it should be noted that there are two main challenges to incorporate QACs into water-dispersed acrylic and melamine paints: leaching from paint films found in wet environments such as bathroom faucet handles due to

the water dissolution of QACs that are attached physically, *e.g.*, 0.014 mM C_{16} -QACs leached from anionic latex paint after immersion in water for 30 days;¹²⁷ and inhibiting the curing of acrylics with melamine due to electrostatic interactions between QACs and negatively charged acrylics.¹³³ Aromatic multicarboxylic acids such as trimesic acid were used to form ionic bonds with the quaternary ammonium groups of QACs to reduce their water solubility, while phosphoric acid (H₃PO₄) was used to neutralize the positive charge of QACs and inhibit electrostatic interactions with acrylic groups.¹³³ The obtained surfaces had improved antiviral activity of >99% against influenza A virus (H1N1) along with greater surface hardness (pencil hardness of H) and solvent resistance.

For glass, textile, and plastic surfaces, either encapsulating QACs in nm-/um-sized capsules or introducing reactive functional moieties to attach QACs can also be performed by pre-coating surfaces with chemicals such as SiO₂ NPs (SNPs) and benzophenone.^{123,135} For example, negatively charged SNPs with 8 nm diameter were coated with DDAB and then the DDAB-coated SNPs were adhered onto a polydopamine-functionalized glass coverslip.¹²³ After 30 min incubation of influenza A virus (H1N1) with a glass surface bearing attached DDAB-coated SNPs, no virus survival was observed, indicating at least 99.9% antiviral activity. By constrast, uncoated glass coverslips had no antiviral activity. In addition, catechol and aryl azide react with functional groups such as hydroxyl, amino, and carboxyl to covalently attach QACs to surfaces such as cellulose and glass,^{136,137} but not to plastics such as polyethylene (PE), PP, and polystyrenes. Alternatively, a benzophenone moiety has been used to coat QACs on plastics since benzophenone is a photocrosslinker that reacts with C-H bonds through hydrogen abstraction by the triplet excited state, forming a C-C covalent bond on surfaces bearing C-H bonds.138 QACs with benzophenone-based esters (BEst) such as 6-(4-benzoylphenoxy)-N-(2-(decyloxy)-2oxoethyl)-N,N-dimethylhexan-1-aminium bromide and benzophenone-based amides (BAm) such as 6-(4-benzoylphenoxy)-N-(2-(decylamino)-2-oxoethyl)-N,N-dimethylhexan-1-aminium bromide have been coated on medically relevant surfaces such as cotton, PU, polyvinyl chloride (PVC), and PP, followed by UV light irradiation (Fig. 5b).¹³⁵ After 30 min exposure of influenza A virus (H1N1) to uncoated PU surfaces, there was a negligible change in viral infectivity compared to the initial viral titre of 4.4×10^4 PFU mL⁻¹, while BEst- and BAm-coated surfaces reduced the viral titre by $>10^3$ - and $>10^4$ -fold, respectively.

QAC-coated antiviral glass and PE materials were coated with *N*-alkylated PEIs such as *N*,*N*-dodecylmethyl PEI and *N*,*N*-hexylmethyl PEI by dipping into an *N*-alkylated PEI solution, followed by drying (Fig. 5c).^{100,125,126,139,140} *N*,*N*-Dodecylmethyl PEIs with molecular weights of 2.17–217 kDa and zwitterionic *N*-(15-carboxypentadecyl)–PEI reduced the viral titre of influenza A virus (H1N1) by > 10⁴-fold within 30 min, while the viral titre reduction was only 10^{0.47}- and 10^{0.025}-fold for anionic *N*-(11-carboxyundecanoyl)–PEI and neutral *N*-(undecanoyl)–PEI, respectively.¹⁰⁰ Hyperbranched QAC polymers coated on glass and plastics reduced the infective dose of enveloped influenza A virus (H1N1) by 10^{5.0}- and 10^{3.5}-fold after 1 h exposure, respectively, while there was less antiviral activity against poliovirus Sabin1 even after 6 h exposure.124 N-Alkylated PEIs noncovalently coated on surfaces were stable in water but leached in organic solvents such as butanol, while N-alkylated PEIs covalently bonded on surfaces did not leach.¹⁴¹ 4-Bromobutyryl chloride and aryl azides such as nitrophenylazide reacted with amino, hydroxyl, or carboxyl to covalently attach QACs to cotton cellulose and glass,^{137,141} while benzophenone was used as a UV photocrosslinker to coat QACs on plastic surfaces such as PE and PP containing C-H groups.¹⁴² For surfaces bearing NH₂ functional groups such as commercial NH2-glass, covalently attached N-alkylated PEI coatings were prepared by N-acylation of surfaces with 4-bromobutyryl chloride, nucleophilic substitution of Br with PEIs, and N-alkylation of PEIs by haloalkyls such as CH₃(CH₂)₆Br and CH₃I (Fig. 5d).¹⁴¹ The obtained glass covalently bonded with branched N,N-hexylmethyl PEIs showed 99.9% antiviral activity to inhibit rotavirus and poliovirus after 0.5 h exposure.¹²⁵ Various surfaces coated by N-alkylated PEIs have antiviral activities against viruses such as influenza A and MS2 bacteriophage (cf. Table 2). The antiviral activity of glass surfaces with a covalently attached N-alkylated PEI coating has no correlation with various properties such as surface charge, hydrophobicity, adhesion strength, surface roughness, and polymer-chain extension length, but the degree of viral titre reduction increased from 10^{0.2}- to 10^{3.5}-fold upon increasing the surface density of quaternary ammonium groups from 0 to 0.12 nmol cm⁻².¹⁰⁸ Surfaces such as cotton, paper, and plastics, including PE, PP, and PVC, can be functionalized with N-alkylated PEIs and are expected to inhibit other enveloped viruses, albeit further testing is warranted to validate these antiviral surface coatings against current viral threats.

3.2. Ag compounds and AgNP-coated surfaces

Ag compounds such as $AgNO_3$, Ag_2O , Ag_2S , Ag^+ -metal organic framework (MOF), and AgNPs have been employed as antiviral materials (Table 3). Glass surfaces coated with AgNO₃, Ag₂O, and Ag₂S reduced the viral titre of influenza A virus (H1N1) by 10^{6.3}-, 10^{6.3}-, and 10^{0.042}-fold after 15 min exposure, while the corresponding reductions were 10^{3.3}-, 10^{1.7}-, and 10^{1.3}-fold for $Q\beta$ bacteriophage after 1 h exposure, respectively.¹¹⁶ These findings indicate greater antiviral activity against enveloped viruses than non-enveloped viruses. More soluble Ag₂O with a solubility limit of 0.025 g L^{-1} had greater activity than Ag₂S, which is insoluble in water, indicating that Ag⁺ is responsible for virus inactivation. Ag⁺-MOF prepared from Ag₂O, 1,3,5-triaza-7phosphaadamantane, and pyromellitic reduced the titre of human adenovirus 36, a non-enveloped virus, by 10⁴-fold after 30 min exposure, although the inactivation mechanism has not yet been explored.¹⁴⁶ The Ag⁺-MOF also showed antimicrobial activity with a minimum inhibitory concentration that was >2-fold lower than that of AgNO₃, while the half maximal inhibitory concentration against normal human dermal fibroblasts was similar, 50.5 and 44.1 μ M, for Ag⁺-MOF and AgNO₃, respectively. Since the surface of AgNPs oxidizes to Ag₂O in water to release Ag⁺, ¹¹⁶ AgNPs can exhibit antiviral activity for viruses such as hepatitis B virus, HIV-1, respiratory syncytial virus, and

influenza A virus (Table 3),¹⁴⁷ with higher antiviral activity than Ag^+ by 12-fold for HIV-1.¹¹⁷ It is assumed that AgNPs release Ag^0 and Ag^+ clusters with higher antiviral activity than Ag^+ although further experimental evidence is needed.¹⁴⁸

It is currently understood that the antiviral activity of Ag⁺ relates to the inhibition of thiol/disulphide (SH/S-S) rearrangement events that are involved in viral membrane fusion. For example, the HIV-1 E glycoprotein consists of a surface glycoprotein (gp120) and a transmembrane glycoprotein (gp41) that together form a trimer on the viral membrane surface.^{144,149} HIV-1 entry requires gp120 attachment to host cell receptors, which induces a structural SH/S-S rearrangement in gp120 that is necessary for envelope fusion^{144,150} (Fig. 6a and b). Because the coordination number (3.1) of amino acids containing SH for Ag^+ is larger than 0.36 and 0.28–0.4 of those containing S–S and non-sulphur moieties, respectively,¹¹¹ Ag⁺ reacts preferentially with the cysteine residues (RSH and RSSR) of envelope proteins and enzymes to produce AgSR and H⁺ through bond cleavage and hydrolysis, resulting in inactivation of enveloped viruses (Fig. 6c).^{109-111,151}

Past studies have demonstrated that AgNPs can attach to HIV-1 virus particles and the degree of interparticle spacing (approximately 28 nm centre-to-centre distance) was similar to the expected spacing between gp120 proteins (22 nm) (Fig. 6d and e¹⁴⁵), suggesting that the exposed sulphur-bearing residues of gp120 proteins are the reaction sites for AgNPs.^{117,145} As a result, SH/S-S dependent membrane fusion is prevented, rendering HIV-1 inactive. Similarly, Ag⁺ reacts with SH and S–S moieties in the E glycoprotein receptor of dengue virus and in the neuraminidase enzyme of influenza A virus to inactivate those enveloped viruses as well, ^{113,114,116,117,152} since SH/S–S exchange processes are also involved in viral infectivity processes. However, non-enveloped viruses such as coxsackie B3 virus have no such proteins or enzymes that readily react with Ag^{+,113}

From a chemical design perspective, solubility and nanoparticle size are also important.^{153,154} With respect to the former point, AgNO₃ and Ag₂O have greater activity than Ag₂S, which is insoluble in water. With respect to the latter point, when a nanoparticle is larger than the diameter of critical protein features (approximately 15 nm), the nanoparticle will not attach to the SH/S-S region near the binding domain, resulting in lower inactivation efficiency. Accordingly, to date, AgNPs with 5-15 nm diameter have been mainly used as antiviral materials.¹⁵⁵ Chitosan coated with 3 wt% AgNPs with 3.5 nm diameter showed higher antiviral activity (78%) for influenza A virus (H1N1) than chitosan coated with AgNPs with 6.5 nm (60%) or with 12.9 nm (38%) diameter,¹⁰³ indicating a moderate effect of the larger surface area of smaller AgNPs. Ag^{0} -loaded $Al_{2}O_{3}$ (Ag/Al_{2}O_{3}) wafers inactivated 10^{6} PFU mL⁻¹ SARS-CoV-1 and baculovirus within 5 min in air, but no antiviral activity was observed in the absence of O2,156 indicating that the inactivation of viruses occurs by reduction of O_2 by Ag^0 , generation of •OH and $O_2^{\bullet-}$, and •OH- and $O_2^{\bullet-}$ -oxidative disruption of viruses (Fig. 6f).¹⁵⁷ A magnetic hybrid colloid (MHC) loaded with 30 nm diameter AgNPs (Ag30/MHC) showed greater antiviral effects for ϕ X174 bacteriophage and murine norovirus (10⁴- and 10⁶-fold reduction in viral titre, respectively)

Table 3 Virus inactivation on inorganic compound-coated antiviral surfaces

Antiviral material	Virus ^{<i>a</i>} $(N_0 [PFU mL^{-1}])^b$	Virus reduction ratio, ^{c} log ₁₀ (N_0/N_t) (exposure time)	Ref.
Ag ⁺ ,Cu ²⁺ ,Zn ²⁺ -coated glass	HIV-1 $(1.66 imes 10^5)$	>99.9%, 5 (20 min)	113
	HSV (1.0×10^6)	>99.9%, 5.5 (4 h)	
	DENV-2 $(3.98 \times 10^{\circ})$	97%, 2.2 (4 h)	
	Coxsackie B3 virus (3.16×10^4)	37%, 0.2 (4 h)	
AgNO ₃ -coated glass	IAV (H1N1) (1.2×10^8)	>99.9%, 6.3 (0.5 h)	116
	Q β bacteriophage (1.5 × 10 ⁹)	99.9%, 3.3 (0.5 h)	
AgNP (5–25 nm)-modified graphene oxide	FCoV $(4.7 \times 10^4)^d$	92%, 1.1 (1 h)	170
Mercantoethanesulfonate-modified AgNPs (3-13 nm)	(9×10) HSV-1 (2.5 × 10 ³)	96%, 1.35 (1 ft) 97%, 1.57 (45 min)	171
AgNPs (4–13 nm) produced by <i>F. oxysporum</i>	HSV-1 (2×10^4)	78% (2 h)	171
	HSV-2 (2×10^4)	57% (2 h)	
	HPV-3 (2×10^4)	78% (2 h)	
Agnes (5–23 nm) produced by C. indicum	HSV-1 $(2 \times 10^{-})$ HSV-2 (2×10^{4})	80% (2 h)	172
	HPV-3 (2×10^{4})	80% (2 h)	
AgNPs (7–20 nm) produced by <i>Alternaria</i> and <i>Phoma</i> species	HSV-1 (2×10^4)	40% (2 h)	172
	HSV-2 (2×10^4)	56% (2 h)	
	HPV-3 (2×10^4)	60% (2 h)	
AgNP (7 nm)-loaded magnetic hybrid colloid (500 nm) (Ag7/MHC)	X174 bacteriophage (10°)	99.8%, 2.7 (6 h)	158
	Adenovirus serotype 2 (10^6)	49.9%, 1 (6 ll)	
Ag15/MHC	X174 bacteriophage (10^6)	99.9%, 3 (6 h)	158
	Murine norovirus (10^6)	90%, 1 (6 h)	
	Adenovirus serotype 2 (10^6)	49.9%, 0.3 (6 h)	
Ag30/MHC	X174 bacteriophage (10°)	>99.9%, 4 (6 h)	158
	Murine norovirus (10°)	> 99.9%, 6 (6 h)	
AgNP (3.5 nm)-coated chitosan	IAV (H1N1) $(10^3)^d$	78% (1 h)	103
AgNP (6.5 nm)-coated chitosan	IAV (H1N1) $(10^3)^d$	60% (1 h)	103
AgNP (12.9 nm)-coated chitosan	IAV $(H1N1)$ $(10^3)^d$	38% (1 h)	103
PEI-(5 layers), AgNP-coated PES membranes	MS2 bacteriophage (4.0×10^8)	>99.9%, 4 (0.5 h)	143
PEI-(5 layers), AgNP-, CuNP-coated PES membranes	MS2 bacteriophage $(4.0 \times 10^{\circ})$	>99.9%, 4.5 (0.5 h)	143
$Ag (0.2-12 \mu m)$ -coated glass $Ag/Al_{2}O_{2}$ wafers	SARS-CoV-1 (10^6)	>99.9% 6 (5 min)	110
ng/m203 waters	Baculovirus (10 ⁶)	>99.9%, 6 (5 min)	100
Ag ₂ O-coated glass	IAV (H1N1) (1.2×10^8)	>99.9%, 6.3 (0.5 h)	116
	Q β bacteriophage (1.5 × 10 ⁹)	98%, 1.7 (0.5 h)	
Ag ₂ S (3–60 μm)-coated glass	IAV (H1N1) $(1.2 \times 10^{\circ})$	9.2%, 0.042 (0.5 h)	116
Ag.S (3-60 um)-coated glass	Qb bacteriophage $(1.5 \times 10^{\circ})$	95.0%, 1.3 (1 h) 55% 0.26 (0.5 h)	110
AgI-coated glass	Oß bacteriophage $(1.5 \times 10^{\circ})$	44%, 0.25 (0.5 h)	118
Ag ⁺ -MOF	Human adenovirus 36 $(10^2)^d$	>99.9%, 4 (0.5 h)	146
Cu alloy	IAV (H1N1) (2×10^6)	>99.9%, 4 (6 h)	52
Cu (42 μ m)-coated Al ₂ O ₃	MS2 bacteriophage (1.2×10^4)	>99.9%, 3.1 (2 h)	105
Cu/Al ₂ O ₃ waters	SARS-COV-1 (10°) Reculowings (10^{6})	>99.9%, 6 (20 min)	156
$CuCl_{2}$ (500 µM)	AIV (H5N3) (10^7)	>99.9%, >5.5 (48 h)	114
CuCl_2 (525 μ M)	FCoV (1.3×10^8)	75%, 0.6 (1 h)	119
$\operatorname{CuCl}_2(250 \ \mu M)$	AIV (H9N2) (10 ⁶)	99%, 2 (1 h)	161
CuCl ₂ (2.1 µmol)-coated glass	Q β bacteriophage (1.5 × 10 ⁹)	43%, 0.24 (0.5 h)	116
CuCl ₂ -coated glass	QB bacteriophage (1.5×10^{7})	44%, 0.25 (0.5 h)	118
CuCl ₂ /zeonte-coated textile	AIV (H5N1) (10) AIV (H5N3) $(10^{7.0})$	>99.9%, 5.2 (10 mm)	114
CuCl (5.3 µM)	FCoV (1.3×10^8)	99%, 2.25 (1 h)	119
CuCl (2.1 µmol)-coated glass	Q β bacteriophage (1.5 \times 10 ⁹)	>99.9%, 6.6 (0.5 h)	118
Cu _x O-coated N95 mask	IAV (H1N1) $(10^{5.7})^d$	>99.9%, 4.6 (0.5 h)	168
Cu O sected nonveyon DD fibers	AIV (H9N2) $(10^{0.2})^{a}$	>99.9%, 5.1 (0.5 h)	167
Cu _x O-coaled holiwoven PP libers	$IAV (H3N2) (10^{7.5})$	99.7%, 2.5 (0.5 II) 97.8% 1.65 (2 min)	167
	Measles virus $(10^{3.67})$	>99.9%, 3.67 (2 min)	
	RSV (10 ⁴)	96.8%, 1.5 (2 min)	
	Parainfluenza virus 3 (10 ⁸)	92.2%, 1.11 (2 min)	
	Adenovirus type 1 (10°)	99.3%, 2.15 (2 min)	
	Cytomegalovirus (10^{-5})	> 99.9%, 4.3 (2 min)	
Cu ₂ O (0.5–5 µm)-coated glass	IAV (H1N1) (1.2×10^8)	>99.9%, 3.7 (0.5 h)	116
	$Q\beta$ bacteriophage (1.5 × 10 ⁹)	>99.9%, 5.8 (0.5 h)	-10
Cu ₂ O (0.5–5 µm)-coated glass	Q β bacteriophage (1.5 \times 10 ⁹)	>99.9%, 5.8 (0.5 h)	118
	T4 bacteriophage (1.5 $ imes$ 10 ⁹)	>99.9%, 5.8 (0.5 h)	

Table 3 (continued)

Antiviral material	$\text{Virus}^a \left(N_0 \; [\text{PFU} \; \text{mL}^{-1}] \right)^b$	Virus reduction ratio, ^{<i>c</i>} $\log_{10}(N_0/N_t)$ (exposure time)	Ref.
Cu_2O (47 µm)-coated Al_2O_3	MS2 bacteriophage (1.6×10^4)	>99.9%, 3.2 (2 h)	105
CuI-coated glass	Qβ bacteriophage (1.5×10^9)	>99.9%, 6.0 (0.5 h)	118
CuI	FCoV (1.3×10^8)	>99.9%, 7 (1 h)	119
CuI	$IAV(H1N1) (10^6)$	>99.9%, > 5 (1 h)	173
Cu ₂ S (4–95 µm)-coated glass	Q β bacteriophage (1.5 \times 10 ⁹)	>99.9%, 4.6 (0.5 h)	118
CuO (0.5-40 µm)-coated glass	IAV (H1N1) (1.2×10^8)	2.3%, 0.01 (0.5 h)	116
	Q β bacteriophage (1.5 \times 10 ⁹)	18.3%, 0.088 (0.5 h)	
CuO (0.5-40 µm)-coated glass	$Q\beta$ bacteriophage (1.5×10^9)	2.7%, 0.012 (0.5 h)	118
CuO (49 μ m)-coated Al ₂ O ₃	MS2 bacteriophage (1.4×10^4)	30.8%, 0.16 (2 h)	105
CuS (0.5-40 µm)-coated glass	Qβ bacteriophage (1.5×10^9)	74.3%, 0.59 (0.5 h)	118
$ZnCl_2$ (10 mM)	IAV(H1N1) (10 ^{4.4})	90%, 2 (1 h)	169
Zn ²⁺ (0.033%)-embedded PA66 fabric	$IAV(H1N1)(10^4)$	90%, 2 (1 h)	169

^{*a*} IAV, AIV, HSV, DENV, HPV, RSV, and FCoV represent influenza A virus, avian influenza virus, herpes simplex virus, dengue virus, human parainfluenza virus, respiratory syncytial virus, and feline coronavirus, respectively. ^{*b*} N_0 is the number of virus particles [PFU mL⁻¹] before exposure unless specified otherwise. ^{*c*} Reduction ratios are calculated from $(1 - N_t/N_0) \times 100\%$ and $\log_{10}N_0 - \log_{10}N_t$ (= $\log_{10}(N_0/N_t)$), where N_t is the number of infectious virus particles after exposure to an antiviral surface for time *t*. ^{*d*} N_0 is the number of virus particles with the unit of TCID₅₀ mL⁻¹ before exposure.



Fig. 6 Ag-mediated inactivation of viral structural proteins. 3D structure of a trimeric S protein on a native HIV-1 viral membrane (a) and gp120 core (red) derived from the complex of antibody b12 (Protein Data Bank ID, 2NY7) with V1 and V2 loops (residues 121-203, yellow) and V3 loop (residues 300-328, green) (b).¹⁴⁴ The white arrow (a) points to the location of gp41. Reaction of Ag⁺ with cysteine residues in viral proteins (c). High angle annular dark field (HAADF) scanning transmission electron microscopy (STEM) images of HIV-1 before (d) and after (e) exposure to AgNPs.¹⁴⁵ Reaction of $^{\bullet}OH$ and $O_2^{\bullet-}$ generated from AgNPs with viral proteins (f).

than MHCs loaded with 7 nm diameter AgNPs (Ag7/MHC; $10^{2.7}$ and 10^3 -fold reductions, respectively) and 15 nm diameter AgNPs (Ag15/MHC; 10- and 10-fold reductions, respectively), due to a higher concentration of released Ag⁺ in solution from Ag30/MHC (400 mg L⁻¹) than from Ag7/MHC (57.5 mg L⁻¹) and Ag15/MHC (275 mg L⁻¹).¹⁵⁸ However, AgNPs/MHCs could not inactivate adenovirus serotype 2, indicating some degree of virus specificity.

3.3. CuNP, Cu alloy, and Cu compound-coated surfaces

While Ag compounds and AgNPs have high antiviral activity among metallic compounds and metal NPs, they have high costs,

which limit practical applications. For this reason, Cu and Cu compound-coated surfaces have received more attention on account of the lower cost of Cu and its broader spectrum of antiviral activity against enveloped and non-enveloped viruses,¹¹⁸ while Ag compounds only inactivate enveloped viruses but not non-enveloped viruses (Table 3). Cu-loaded Al_2O_3 (Cu/Al₂O₃) wafers needed 15 min longer than Ag/Al₂O₃ wafers (5 min) to completely inactivate 10⁶ PFU mL⁻¹ SARS-CoV-1 and baculovirus in air, and no antiviral activity was observed in the absence of O₂.¹⁵⁶ Cu alloy and cartridge brass surfaces with 70% Cu inactivated 10³ PFU mL⁻¹ human coronavirus 229E within 5 min by releasing Cu⁺ and Cu²⁺ to induce disintegration of the viral envelope.⁴²

The activity was quenched by the presence of ethylenediaminetetraacetic acid and bathocuproine disulfonate, which function as chelators of Cu^{2+} and Cu^{+} , respectively, with exposure for > 2 h.

Within this scope, the inactivation of coronaviruses on Cu surfaces is considered to occur initially by released Cu^{2+} , and then by released Cu^+ to generate •OH and $O_2^{\bullet-}$ from the redox reaction of water and $O_2^{\bullet,2}$ Virus exposure to various reactive species such as Cu^{2+} , Cu^+ , •OH, and $O_2^{\bullet-}$ results in the oxidation of membrane proteins and/or nucleoproteins. Although the exact inactivation mechanism is not clear, the primary interaction sites may be located on key proteins associated with viral envelopes and capsids. It has been suggested that cysteine amino acids interact with Cu(111) *via* the S atom of SH and the two O atoms of COO^- functional groups *in vacuo*,¹⁵⁹ which could explain the interaction of viral proteins and enzymes with Cu surfaces (Fig. 7a). In addition, Cu^{2+} and Cu^+ can disrupt viruses through disrupting SH/S–S exchange of proteins *via* copper-catalysed oxidation.¹⁶⁰

Antiviral activity against various viruses has been reported for Cu compounds (Table 3).^{105,116,118,119} In general, cuprous compounds such as CuCl, Cu₂O, CuI, and Cu₂S have greater antiviral activity than cupric compounds such as CuCl₂, CuO and CuS.^{114,116,118,119,161} Although the CuCl solubility limit of 0.0024 M is lower than the CuCl₂ solubility limit of 5.43 M in water, Cu⁺ has greater antiviral activity than Cu²⁺. For example, the viral titre of Q β bacteriophage decreased by 10^{6.6}-fold after 30 min incubation on CuCl-coated glass, while the viral titre reduction was only 10^{0.24}-fold on CuCl₂-coated glass. Cu⁺ from CuCl at 5.3 µM inactivated feline coronavirus with a viral titre reduction of 10^{3.3}-fold,¹¹⁹ while Cu²⁺ from CuCl₂ at 250 and $525 \,\mu$ M concentrations inactivated avian influenza virus (H9N2) and feline coronavirus with viral titre reductions of 100- and 4-fold, respectively, after 1 h exposure.^{119,161}

Mechanistically, Cu²⁺ either oxidizes RSH to produce RSSR or Cu⁺/RS⁻ complex (Cu(SR)₂), and H⁺, or acts as an activator to accelerate the hydrolytic breakage of S-S bonds, 109-112 disrupting viral proteins and enzymes on lipid envelopes (Fig. 7b).¹⁶² Cu²⁺ interacts with viral nucleic acids, such as double-stranded DNA (dsDNA) at a stability constant of 2.4×10^4 M⁻¹ through bonding to nucleotides and base pairs,^{113,114,163} causing degradation of the helical structure and inactivating dsDNA viruses such as herpes simplex virus.¹¹³ Since Cu⁺ interacts with cysteine amino acids and has high affinity for the S atom of SH functional groups with 1.2 stoichiometry and a 7.9 \times 10⁹ M⁻¹ stability constant, there are two possible binding configurations involving Cu⁺ and S atom binding alone or in conjunction with Cu⁺ binding to an N atom of NH₂ functional groups.^{122,164} Hence, Cu⁺ can react with the SH and NH₂ groups of cysteine residues in capsid and envelope proteins (Fig. 7c). A representative example of Cu²⁺ inactivating a disulphide bond in a SARS-CoV-2 S protein is presented in Fig. 7d and demonstrates the types of protein structural elements with which Cu²⁺ can interfere.¹⁶⁵

In addition, Cu⁺ induces the redox reaction of water and O₂ to generate •OH and O₂•⁻ for inactivating viruses through oxidative disruption.¹¹⁹ Cu⁺ initially reduces O₂ to produce Cu²⁺ and O₂•⁻, followed by the formation of H₂O₂ from the redox reaction of O₂•⁻ with water and the generation of •OH from the reaction between Cu⁺ and H₂O₂. Since •OH reacts with cysteine, tryptophan, and tyrosine residues at rates of $(1.3–3.5) \times 10^{10} \text{ M}^{-1} \text{ s}^{-1}$ that are faster than the rates of 1.7×10^7 –8.5 × $10^9 \text{ M}^{-1} \text{ s}^{-1}$ for



Fig. 7 Virus inactivation on surfaces coated with a Cu alloy or NPs (a), $CuCl_2$ (b), or cuprous compounds such as CuCl, Cu_2O , and CuI (c). Illustrative example of Cu^{2+} interaction with a disulphide bond in the S protein⁴⁸ of SARS-CoV-2 (d).

other amino acids,¹⁶⁶ the initial oxidation is suggested to occur on those particular amino acids found within viral proteins and enzymes.

A Cu_rO-coated filter inactivated influenza A virus (H1N1) with a viral titre reduction of 59-fold after 40 min filtration,¹⁶⁷ and a CurO-coated face mask inactivated influenza A virus (H1N1) and avian influenza virus (H9N2) by 10³-fold after 30 min filtration,¹⁶⁸ although the oxidation states of Cu were not reported. Cuprous compounds such as Cu₂O and Cu₂S reduced the viral titre by 10^{3.7}-, 10^{3.2}-, and 10^{5.8}-10⁶-fold for influenza A virus (H1N1),¹¹⁶ MS2 bacteriophage,¹⁰⁵ and Qβ bacteriophage,¹¹⁸ at 0.5, 2, and 0.5 h, respectively, while cupric compounds such as CuO and CuS had no or less activity with viral titre reductions around only 10^{0.01}-10^{0.59}-fold (Table 3).105,116,118,119 The lower activity of CuO and CuS is due to the lower activity of Cu²⁺ compared to that of Cu⁺.¹¹⁹ Cu_2O (2.1 µmol)-coated glass released Cu^{2+} and Cu^+ at 21.6 and 7.19 µM concentrations, respectively, and yielded negligible inactivation of $Q\beta$ bacteriophage with a viral titre reduction of only 10^{0.2}-fold.¹¹⁸ When filter paper with 105 µm thickness and 30 nm pore size was placed on Cu₂O- or Cu₂O-modified by 1H-benzotriazole-coated glass, the viral titre reduction of $Q\beta$ bacteriophage decreased from around $10^{5.8}$ - to $10^{1.2}$ - and 10^{0.6}-fold, respectively,¹¹⁸ which supports that the direct interaction between cuprous compounds and QB bacteriophage is important for virus inactivation. An 8 h exposure of bovine serum albumin protein led to strong adsorption on Cu2O-coated, but not CuO-coated, surfaces and 1 h incubation of alkaline phosphatase enzyme in Cu₂O suspensions led to a 70% decrease in the enzymatic activity via denaturation, including 30% adsorption-related loss, while there was no change in enzymatic activity upon incubation in CuO suspensions.¹¹⁸ The interaction of Cu₂O with viral capsid proteins through the S atom of SH and N atom of NH₂ groups in cysteine residues¹²² was suggested to cause the disruption of viral proteins and enzymes by Cu⁺ released from Cu₂O (cf. Fig. 7c).¹¹⁶ CuI at 1000 mg L^{-1} (5.26 mM) in water released Cu^{2+} and Cu^{+} at 426.4 and 3.6 μM concentrations, respectively, which resulted in the inactivation of feline calicivirus with a viral titre reduction of 10^{6.2}-fold after 1 h exposure.¹¹⁹ Comparably, the viral titre reduction of feline calicivirus was 4- and 10^{3.3}-fold upon treatment with CuCl₂ and CuCl at 525 and 5.3 µM, respectively. Together, these data suggest that the contact of feline calicivirus with CuI surfaces is better for virus inactivation. Since the Trp383 and Met651 residues in the capsid protein of feline calicivirus were oxidized to tryptophan-OH and methionine sulfoxide, respectively, the inactivation corresponds to oxidative disruption by Cu⁺ in spite of its two-orders of magnitude lower concentration than that of Cu²⁺.¹¹⁹

In addition to Ag and Cu compounds, 10 mM ZnCl₂ inactivated influenza A virus (H1N1), resulting in a viral titre reduction of 99% after 1 h exposure.¹⁶⁹ It was found that ZnCl₂ reduced 30% of haemagglutinin proteins, while there was no effect on neuraminidase enzymes or viral RNA. These findings implied that Zn²⁺ can inactivate influenza A virus (H1N1) through the destabilization of surface proteins rather than internal proteins or nucleic acids. Zn²⁺ (0.033%)-embedded polyamide 6.6 (PA6.6) fabric reduced the

viral titre of influenza A virus (H1N1) by a 10^2 -fold higher extent than a PA66 control fabric. Interestingly, a glass surface coated with AgNO₃, Cu(NO₃)₂, and Zn(NO₃)₂ resulted in viral titre reductions of 99.9% after 20 min exposure of HIV-1, while the corresponding reduction levels were 99.9, 97, 77, and 37% for herpes simplex virus type 1, dengue virus type 2, influenza A virus (H1N1), and coxsackie B3 virus, respectively,¹¹³ which indicated different sensitivities of viruses to Ag⁺, Cu²⁺, and Zn²⁺.

As noted above, there are many types of chemically functionalized surfaces for virus inactivation. Surfaces coated by QACs such as DDAB and N-alkylated PEIs exhibit antiviral activity against enveloped viruses, and QACs covalently bonded on surfaces have non-leaching properties (Table 3). The antiviral activity of QACs decreases with increasing the operating time due to the accumulation of inactivated viruses on surfaces. On the other hand, the antiviral activity exhibited by AgNPs, CuNPs, and Ag, Cu, and Zn compounds is mainly due to released metal ions on the surface, which occurs through different mechanisms. AgNPs have greater antiviral activity against enveloped viruses than CuNPs, but the high cost of Ag limits practical application along with ineffectiveness against non-enveloped viruses. On the other hand, Cu-coated surfaces have received more attention due to the lower cost of Cu and since Cu and cuprous compounds such as Cu₂O, CuI, and CuCl can exhibit antiviral activity against both enveloped and non-enveloped viruses.

4. Light-induced surfaces for virus inactivation

While photocatalysts and triplet photosensitizers such as TiO₂ and porphyrins have no antiviral activity in the dark, they can inactivate viruses under light irradiation through photocatalysis and triplet photosensitized reactions, respectively. The photocatalytic disruption of viruses is generally considered to proceed through oxidation by the photogenerated holes of photocatalysts, and oxidation by oxidative reactive oxygen species (ROS) such as $^{\bullet}$ OH and $O_2^{\bullet-}$ that are generated through the redox reaction of water and O2 with photogenerated holes and electrons.¹⁷⁴ Triplet photosensitized disruption of viruses proceeds through oxidation by ${}^{1}O_{2}$ that is generated by triplet-triplet energy transfer from triplet photosensitizers to O2.¹⁷⁵ Such ROS react initially with the amino acids of proteins found in lipid envelope and capsid structures to inactivate viruses depending on the virus type. Light-induced virus inactivation can be divided into two classes depending on whether UV or visible light irradiation is required to render the surface functional. The corresponding mechanisms are discussed below.

4.1. UV light-driven TiO₂ photocatalyst-coated surfaces

Photocatalytic disruption of viruses^{101,200,201} is considered to proceed through oxidation by photogenerated holes on photocatalyst surfaces, and by ROS such as •OH and $O_2^{\bullet-}$ generated from the redox reaction of water and O_2 with photogenerated holes and electrons under light irradiation (Fig. 8a and b).^{202,203} Oxidative disruption of lipid envelopes and protein capsids by





photogenerated holes occurs on the photocatalyst surface through hole transfer, and even in air by ROS that diffuse from the surface.^{201,204} The degree of virus-inactivating activity increases with a greater amount of tyrosine residues in viral proteins, since tyrosine has the highest $E_{\text{HOMO}} = -8.81 \text{ eV}^{201}$ and lowest $E_{ox} = 0.97$ V vs. NHE at pH = 7^{205} among amino acids found in envelope and capsid proteins. Coating TiO₂ on membranes, filters, and films has also been shown to inactivate bacteriophages such as MS2 and OB and mammalian viruses such as respiratory syndrome virus and influenza virus that are attached to surfaces in an air environment under UV light irradiation (Table 4). For example, coating TiO₂ on a cell culture plate reduced the viral titre of avian influenza virus (H9N2) from 10^4 to $10^{2.72}$ PFU mL⁻¹ under UV light irradiation at 0.5 mW cm⁻² for 1.5 h.¹⁷⁶ A titanium apatite photocatalyst-coated filter inactivated SARS-CoV-1 with an initial infectivity of $10^{7.5}$ PFU mL⁻¹ under UV light irradiation at 1 mW cm⁻² for 6 h.¹⁷⁷ A TiO₂-coated aluminium filter inactivated influenza A virus (H1N1) by reducing aerosolassociated infectivity of (1.2–2.6) \times $10^{6}~\text{PFU}~\text{mL}^{-1}$ to an undetectable level of infectivity ($<300 \text{ PFU mL}^{-1}$) under 375 nm light irradiation at 10 mW cm⁻² for 7 min.¹⁷⁸

A high concentration of photogenerated holes and ROS as well as close contact between viruses and TiO₂ can increase the level of photocatalytic inactivating activity. Anatase TiO₂ doped with electron acceptors such as metals and semiconductors as well as fluorinated TiO₂ inactivated viruses with higher activity than TiO₂, due to enhancing photogenerated charge separation and oxidation of H₂O and OH⁻ by photogenerated holes (see Fig. 8c and d).^{206–209} For example, TiO₂ surfaces with 2 nm diameter AgNP deposits inactivated MS2 bacteriophage at a 5.8-fold quicker rate than TiO₂ alone under UV light irradiation at 2.5 mW cm⁻² for 2 min (Table 4).¹⁸⁰ The degree of

photocatalytic virus-inactivating activity decreased by 104.2- $10^{4.7}$ -fold in the presence of methanol and *t*-butanol, which acted as scavengers of photogenerated holes and •OH, suggesting that both species are involved.¹⁸⁰ A mixture of anatase and rutile TiO₂ with a ratio of 7:3 reduced the viral titre of MS2 bacteriophage by 10^{1.9}-fold after 1.5 min exposure, which was higher than the results achieved using pure anatase and rutile TiO_2 that yielded $10^{1.0}$ and $10^{0.75}$ -fold reductions, respectively, suggesting enhanced charge separation at the interface between anatase and rutile TiO₂.¹⁸⁴ Fluorination of TiO₂ surfaces by adsorption of F⁻ inactivated MS2 bacteriophage by a twoorders of magnitude greater degree than TiO₂ under UV light irradiation at 10 µW cm⁻² for 1 h.¹⁸² TiO₂ (P25) with SNP deposits reduced the viral titre of MS2 bacteriophage by 10^{5.5}fold after 1.5 min exposure, which was four-orders of magnitude greater than the 10^{1.5}-fold reduction level achieved using TiO₂ under UV-A light (315–400 nm) irradiation at 2.5 mW cm⁻².¹⁸³ The adsorption of MS2 bacteriophage increased by 37-fold for TiO_2 surfaces with SNP deposits, ¹⁸³ suggesting that the interface between negatively charged SNPs and positively charged TiO₂ enhances adsorption due to charge heterogeneity of MS2 bacteriophages²¹⁰ and that direct oxidation of adsorbed MS2 bacteriophage by photogenerated holes is the initial process during photocatalysis.

4.2. Visible light-driven photocatalyst-coated surfaces

Due to its wide band gap of $\geq 3.0 \text{ eV}$,²¹¹ TiO₂ exhibits photocatalytic virus-inactivating activity under UV light irradiation. Visible light (400–700 nm)-driven TiO₂ doped with metal ions,¹⁸⁶ metal oxides,¹⁸⁵ and non-metal anions,¹⁸⁸ as well as other visiblelight-responsive semiconductors such as g-C₃N₄,^{187,189} WO₃,^{190–192} BiWO₄,¹⁸⁷ and Ag/AgCl¹⁸⁷ inactivated viruses under visible light

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 Table 4
 Virus inactivation on photocatalyst- and triplet photosensitizer-coated surfaces

Photocatalyst- or triplet photosensitizer-coated surface	Virus ^a (N_0 [PFU mL ⁻¹]) ^b	Light source, wavelength, power	Virus reduction ratio, ^{<i>c</i>} $log_{10}(N_0/N_t)$ (exposure time)	Ref.
TiO ₂ -coated glass	IAV (H1N1) (10 ⁸)	Black light lamp, 300–425 nm, 0.1 mW cm^{-2}	>99.9%, 4 (8 h)	102
TiO ₂ -coated cell culture plate	AIV $(H9N2)$ (10^4)	Black light lamp, 365 nm, 0.5 mW cm ^{-2}	99.8%, 2.72 (1.5 h)	176
Titanium apatite-coated filter	SARS-CoV-1 $(10^{7.5})$	UV lamp, 365 nm, 1 mW cm ^{-2}	>99.9%, 4 (6 h)	177
TiO ₂ -coated aluminium filter	IAV (H1N1) $((1.2-2.6) \times 10^6)$	UV-LED lamp, 375 nm, 10 mW cm $^{-2}$	>99.9%, 3.9 (7 min)	178
TiO_2 (P25)-coated ceramic membrane	MS2 bacteriophage (10^5)	UV tubes, 253 nm, 1.7 mW cm^{-2}	99.9%. 3 (0.25 h)	179
2 ()	Φ X174 bacteriophage (10 ⁵) PR772 bacteriophage (10 ⁵)		99.9%, 3 (5.8 h) 99.9%, 3 (>20 h)	
TiO ₂ (P25)-coated glass 5.95%Ag/TiO ₂ (P25)-coated glass Anatase TiO ₂ -coated glass	MS2 bacteriophage (7×10^9)	UV-A lamp, 315–400 nm, 2.5 mW cm ⁻²	>99.9%, 4.0 (5.1 min) >99.9%, 5.4 (1.5 min) >99.9%, 4.0 (16.7 min)	180
3.94% Ag/anatase TiO ₂ -coated glass Cu/TiO ₂ $(1:5.5)$ non-woven fabric	HNV (10 ^{6.7} genomic	UV-A LED, 365 nm, 5000 mW $\rm cm^{-2}$	99.8%, 2.8 (1.5 min) >99.9%, 3.8 (1 h)	181
TiO (DOC) spated along	copies m^{-3})	Elucroscopt lown $265 \text{ pm} = 10 \text{ wW} \text{ cm}^{-2}$	05 00/ 0.05 (1 b)	100
110 ₂ (P25)-coated glass	Feline calcivirus (10 ⁸) Murine norovirus (10 ^{8.9})	Fluorescent lamp, 365 nm, 10 µw cm	85.9%, 0.85 (1 h) 80.9%, 0.72 (1 h) 70.5%, 0.53 (1 h)	182
Fluorinated TiO_2 (P25)-coated glass	MS2 bacteriophage (10^{10}) Feline calcivirus (10^8)	Fluorescent lamp, 365 nm, 10 μ W cm ⁻²	99.8%, 2.6 (1 h) 99%, 2.0 (1 h)	182
TiO ₂ (P25)-coated glass 5% SiO ₂ (SNPs)/TiO ₂ (P25)-coated glass	Murine norovirus (10^{-5}) MS2 bacteriophage (3×10^7)	UV-A lamp, 350 nm, 2.5 mW cm^{-2}	99.8%, 2.6 (1 h) 98.4%, 1.8 (1.5 min)	183
Anatase TiO_2 -coated glass Rutile TiO_2 -coated glass	MS2 bacteriophage (10^9)	Black light lamp, 300–425 nm, 2.2 mW $\rm cm^{-2}$	90%, 1.0 (1.5 min) 82.2%, 0.75 (1.5 min)	184
Mixed-phase TiO ₂ - coated glass 0.25% Cu _x O(Cu ⁺ /Cu ²⁺ = 0.13)/TiO ₂ -coated plass	Q\beta bacteriophage (1.2 \times $10^{11})$	White light bulbs, >400 nm, 800 lux $(0.12 \text{ mW cm}^{-2})$	98.7%, 1.9 (1.5 min) >99.9%, 7.5 (40 min)	185
0.25% Cu ²⁺ /TiO ₂ -coated glass		(012)	99.2%, 2.1 (1 h)	
Cu^{2+}/TiO_2 -coated nanofiber (Cu:Ti = 1:7)	f2 bacteriophage (10 ⁵)	Xenon lamp, >400 nm, 100 mW cm ⁻²	>99.9%, 5 (1 h)	186
$TiO_{2-x}N_x$ -coated glass	MS2 bacteriophage (10^8)	Xenon-arc lamp, >400 nm, 150 mW cm ⁻²	90%, 1 (5 h)	187
PdO/TiO _{2-x} N _x nanofiber (Pd·N·Ti = 0.03 · 0.1 · 1)	Coliphage MS2 (3×10^8)	Xenon-arc lamp, >400 nm, 100 mW ⁻²	95.0%, 1.3 (1 h)	188
Bi_2WO_6 -coated glass Ag/AgCl-coated glass	MS2 bacteriophage (10^8)	Xenon-arc lamp, >400 nm, 150 mW cm ^{-2}	>99.9%, 4 (5 h) >99.9%, 8 (5 h)	187
g-C ₃ N ₄ -coated glass			>99.9%, 7.5 (5 h)	
$g-C_3N_4$	f2 bacteriophage (3×10^6)	Xenon lamp, >400 nm, 30 mW cm ⁻²	>99.9%, 3.5 (80 min)	189
Ag_3PO_4 (200 nm)			>99.9%, 4.5 (80 min)	
$Ag_3PO_4/g-C_3N_4$			>99.9%, 6 (80 min)	
Graphene/WO ₃ -coated glass	MS2 bacteriophage (2×10^6)	Mercury lamp, >400 nm, 110 mW cm ⁻²	>99.9%, 5 (3 h)	190
Pt/WO ₃ -coated PE terephthalate film	AIV (H1N1) $(10^{4.7})^a$ Q β bacteriophage (10^9)	LED, 410–750 nm, 150 lux $(22 \ \mu W \ cm^{-2})^e$	>99.9%, 3.2 (2 h) >99.9%, 4 (2 h)	191
	AIV (H1N1) $(10^{4.8})^{d}$ Goose parvovirus $(10^{4.6})^{d}$	LED, 410–750 nm, 1000 lux $(0.15 \text{ mW cm}^{-2})^{e}$	>99.9%, 3.3 (2 h) 99.7%, 2.5 (2 h)	
Pt/WO ₃ -coated glass	AIV (H1N1) $(10^7)^{d}$	Fluorescent lamp, >410 nm, 1000 lux (0.15 mW cm ⁻²) ^{f}	>99.9%, 5.5 (2 h)	192
DIMPy-BODIPY	$ ext{DENV-1} (6.5 imes 10^6)^e ext{VSV} (10^7)$	Xenophot bulb, 400–700 nm, 65 mW cm $^{-2}$	>99.9%, 4 (0.5 h) >99.9%, 5 (0.5 h)	193
	HAdV-5 (6.5×10^5)		99%, 2 (0.5 h)	
DIMPy-BODIPY-coated PAN nanofibers TMPyP	VSV (10^6) MS2 bacteriophage $(10^{4.4})$	Xenophot bulb, 400–700 nm, 65 mW cm ^{-2} UV lamp, 365 nm, 2.2 mW cm ^{-2}	>99.9%, 4 (1 h) >99.9%, 4.1 (10 min)	194 195
TMPyP-coated PAN nanofibers	Hepatitis A Virus $(10^{4.4})$ VSV (5×10^8)	Xenophot bulb, 400–700 nm, 65 mW cm^{-2}	>99.9%, 3.7 (10 min) >99.9%, 7 (0.5 h)	101
Benzophenone-polyphenol-coated PVA-co-PE	T7 bacteriophage (10^5)	D65 Daylight lamp, 300–800 nm, 6.5 mW cm^{-2}	>99.9%, 5 (0.5 h) >99.9%, 5 (5 min)	196
nanofibrous membranes				
Buckminster fullerene (C ₆₀)	Semliki forest virus $(10^9)^d$	Mercury short-arc lamp, 495 nm, 2×10^5 lux (20.4 mW cm ⁻²)	>99.9%, 7 (6 h)	197
C ₆₀ /amine-modified-SiO ₂ -coated stainless steel mesh	MS2 bacteriophage (10 ⁵)	Blue LED, 470 nm, 18.8 mW cm ^{-2}	97.8%, 1.6 (1.5 h)	198

^{*a*} IAV, AIV, DENV, VSV, HAdV-5 and HNV represent influenza A virus, avian influenza virus, dengue virus, vesicular stomatitis virus, human adenovirus-5, and human norovirus respectively. ^{*b*} N_0 is the number of virus particles [PFU mL⁻¹] before exposure unless specified otherwise. ^{*c*} Reduction ratios are calculated from $(1 - N_t/N_0) \times 100\%$ and $\log_{10}N_0 - \log_{10}N_t$ (= $\log_{10}(N_0/N_t)$), where N_t is the number of infectious virus particles after exposure to an antiviral surface for time *t*. ^{*d*} N_0 is the number of virus particles with the unit of TCID₅₀ mL⁻¹ before exposure. ^{*e*} N_0 is the number of virus particles with the unit of lux is converted to that with the unit of μ W cm⁻² by 1 lux = 0.146 μ W cm⁻².¹⁹⁹



Fig. 9 Virus inactivation on surfaces coated with visible light-driven photocatalysts such as Cu^{2+}/TiO_2 , $TiO_{2-x}N_x$, $g-C_3N_4$, and WO_3 under visible light irradiation.

irradiation using daylight sources such as a D65 standard daylight lamp, indoor lamps such as white fluorescent/incandescent lamps, white light emitting diodes (LEDs), and a Xenon lamp with filters for sunlight or visible light (Table 4 and Fig. 9). Although visiblelight-responsive TiO₂ photocatalysts are known to inactivate viruses, the effects of many related aspects such as the inhibitory mechanisms, reactive species involved, reaction sites, temperature effects, and so forth are still being unravelled. Substituting Ti⁴⁺ and O²⁻ with metal ions such as Cu²⁺ and anions such as N and S in the crystal lattice of TiO₂ generates impurity energy levels below the conduction band (CB) and above the valence band (VB) of TiO₂, respectively, causing visible light absorption and changing the redox properties.^{212–215} Substituting Ti⁴⁺ with metal ions such as Cu²⁺ in the crystal lattice of TiO₂ generates impurity energy levels below the CB of TiO₂, causing visible light absorption and inactivating viruses. TiO_2 nanofibers doped with Cu^{2+} (Cu^{2+}/TiO_2) were prepared by in situ incorporation via electrospinning of a precursor solution comprising tetrabutyl titanate, cupric nitrate, acetic acid, PVP, and anhydrous alcohol, and post-treatment of the as-spun nanofibers in a furnace to facilitate thermal decomposition of PVP at 450 °C in air, to have a diameter of 100 nm, band gap of 2.30 eV, and Cu²⁺ doped into the crystal lattice of anatase TiO_2 at the atomic ratio of Cu:Ti = 1:7, which caused inactivation of f2 bacteriophage with a viral titre reduction of 105-fold under visible light irradiation at 100 mW cm⁻² for 4 h.¹⁸⁶ TiO₂ doped with Cu_xO (0.25% Cu_xO/ TiO₂, Cu⁺/Cu²⁺ = 0.13) inactivated Q β bacteriophage with a viral titre reduction of 107.5-fold under visible light irradiation at 0.12 mW cm⁻² for 40 min.¹⁸⁵ Substituting O²⁻ in TiO₂ with N anions in the crystal lattice of TiO_2 (N-doped TiO_2 , $TiO_{2-x}N_x$) generates impurity energy levels above the VB of TiO2,216 causing visible light absorption and inactivating MS2 bacteriophage with a viral titre reduction of 10-fold under visible light irradiation at 150 mW cm⁻² for 5 h.¹⁸⁸ PdO-modified $TiO_{2-x}N_x$ $(PdO/TiO_{2-x}N_x)$ with Pd:N:Ti = 0.03:0.1:1 was deposited on a mesoporous activated carbon fibre template by a sol-gel process along with post-calcination in air at 450 °C for 3 h,

which led to inactivation of MS2 bacteriophage with a viral titre reduction of $10^{1.3}$ -fold under visible light irradiation at 100 mW cm⁻² for 1 h.¹⁸⁸

In addition to various visible-light-responsive TiO₂ photocatalysts, g-C₃N₄, WO₃, BiWO₄, and Ag/AgCl with E_g of 2.6-2.7 eV^{211,217–219} have been reported to inactivate viruses such as avian influenza virus, goose parvovirus, and MS2 and QB bacteriophages.^{101,187,190,194,196} The photocatalytic inactivating activity of g-C₃N₄ caused a reduction in viral titre by $> 10^{7.5}$ fold for MS2 bacteriophage under visible light irradiation at 150 mW cm⁻² for 5 h, which is close to the 10⁸-fold reduction induced by Ag/AgCl and higher than $> 10^{4.2}$ - and 10-fold reductions for BiWO₄ and TiO_{2-x}N_x, respectively.¹⁸⁷ g-C₃N₄ containing deposits of Ag₃PO₄ ($E_{\rm g} \approx 2.4$ eV, $E_{\rm VB} \approx 2.9$ V)²²⁰ with 200 nm diameter (Ag₃PO₄/g-C₃N₄) inactivated MS2 bacteriophage with a viral titre reduction of 10^{6.5}-fold under visible light irradiation at 30 mW cm⁻² for 80 min, which was higher than 10^{3.5}- and 10^{4.5}-fold reduction levels achieved with g-C₃N₄ and Ag₃PO₄, respectively.¹⁸⁹ Glass coated with a 80 nm thick film that included WO₃ (15 nm) and graphene (0.9 nm) with a W/C atomic ratio of 1.2 increased viral titre reduction of MS2 bacteriophage by 10⁵-fold compared to 10^{0.8}-fold for WO₃coated glass under visible light irradiation at 110 mW cm⁻² for 3 h.¹⁹⁰ Coating materials with WO₃ is now being practically used as a visible-light-response photocatalyst to functionalize high-touch car surfaces with antiviral and antibacterial activity. Glass and polyethylene terephthalate films coated by a composite of WO₃ (100-200 nm) and Pt NPs (3-20 nm) with a Pt weight ratio of 0.03-5% inactivated avian influenza virus (H1N1) and goose parvovirus with viral titre reductions of 10^{3.3}-10^{5.5} and 10^{2.5}-fold, respectively, under visible light irradiation at 0.15 mW cm⁻² for 2 h using a conventional room lamp.^{191,192} Although further mechanistic investigation is warranted, it was suggested that generation of $O_2{}^{\bullet-}$ and H_2O_2 from the reduction of O2 in the presence of H2O by photogenerated electrons, and hot electrons by surface plasmon resonance (SPR) excitation of AgNPs provided the reactive species for g-C₃N₄ ($E_{CB} = -1.3$ V),²²¹ WO₃ ($E_{CB} = 0.5$ V),^{222,223} and Ag/AgCl,²¹⁹ respectively.

TiO₂ is mostly used as a UV photocatalyst to inactivate viruses under UV light irradiation at 0.01–3 mW cm⁻². On the other hand, various materials have been studied as visible-light-responsive photocatalysts to inactivate viruses at 100–150 mW cm⁻². Interestingly, Cu_xO/TiO₂ (Cu⁺/Cu²⁺ = 0.13) and Pt/WO₃ inactivate viruses under visible light irradiation using white incandescent lamps and white LEDs at 0.022–0.15 mW cm⁻².

4.3. Visible light-driven triplet photosensitizer-coated surfaces

Triplet photosensitizers such as benzophenones, porphyrins, boron dipyrromethenes (BODIPY), and fullerene (C₆₀) can generate ¹O₂ through triplet-triplet energy transfer from tripletexcited photosensitizers to O₂ with $E(^{1}\Sigma - ^{3}\Sigma) = 1.63 \text{ eV}^{175,224,225}$ to inactivate viruses under light irradiation (Table 4 and Fig. 10). For example, 5,10,15,20-tetrakis(1-methylpyridinium-4yl)porphyrin (TMPyP) (λ_{max} = 441 nm (2.81 eV), $E_{triplet}$ = 1.88 eV)²²⁶ and 2,6-diiodo-1,3,5,7-tetramethyl-8-(N-methyl-4-pyridyl)-4,4'difluoroboradiazaindacene (DIMPy-BODIPY) ($\lambda_{max} = 546$ nm (2.27 eV), $E_{\text{triplet}} = 2.10 \text{ eV}$)¹⁹³ were embedded by electrospinning into polyacrylonitrile (PAN) nanofibers to inactivate viruses such as vesicular stomatitis virus and human adenovirus-5 with viral titre reductions of 10⁵- and 10⁴-fold, respectively, under visible light irradiation at 65 mW cm⁻² for 1 and 0.5 h.¹⁹³ Poly(vinyl alcohol-co-ethylene) (PVA-co-PE) nanofiber membranes prepared by electrospinning, followed by covalent bonding of benzophenones and polyphenols as triplet photosensitizers ($\lambda_{max} = 332 \text{ nm}$ (3.73 eV)) inactivated T7 bacteriophage (a non-enveloped virus) with a viral titre reduction of 105-fold under daylight irradiation (300–800 nm) at 6.5 mW cm⁻² for 5 min.¹⁹⁶ C₆₀ (λ_{onset} = 661 and 408 nm (1.88 and 3.04 eV, respectively), $E_{\text{triplet}} = 1.5-2.2 \text{ eV}$ inactivated Semliki forest virus and vesicular stomatitis virus with viral titre reductions of around 10⁷-fold in both cases under 495 nm light irradiation at 29.4 mW cm⁻² for 6 h,¹⁹⁷ while the reduction was 101.7-fold for MS2 bacteriophage on a stainless steel mesh coated with composites of C60 and amine-modified SiO2 under 470 nm light irradiation at 18.8 mW cm⁻² for 1.5 h.¹⁹⁸

Triplet photosensitizers such as benzophenones, porphyrins, and BODIPY are damaged through photobleaching by oxidative reactions with ${}^{1}O_{2}$ that is generated during triplet photosensitized reactions. It is noted that C_{60} shows higher photostability with a weaker absorption coefficient of 710 M⁻¹ cm⁻¹ at 536 nm²²⁸ than $(1-5) \times 10^{5}$ and $(4-10) \times 10^{4}$ M⁻¹ cm⁻¹ at 400–450 nm and 530–570 nm for porphyrins²³⁰ and BODIPY,²³¹ respectively. Together, these results outline key design parameters, including strong absorption coefficient, high yield of the triplet-state and ${}^{1}O_{2}$ generation, and high photostability, that are required for visible-light-responsive triplet photosensitizers to exhibit high levels of virus-inactivating activity.

5. SARS-CoV-2 inactivation

SARS-CoV-2 is an enveloped virus with a single positivestranded RNA genome. Current findings support that SARS-CoV-2 is highly stable at 4 °C with only a 10^{0.7}-fold reduction of viral titre after 14 days at that temperature.⁷⁵ Such findings highlight the critical need for disinfecting surfaces to mitigate potential spreading of SARS-CoV-2. Both UV irradiation and chemicals such as hypochlorite, H₂O₂, ethanol, and a commercial QAC have been demonstrated to reduce the viral titre of SARS-CoV-2 by 10³-10⁵-fold.⁸⁶⁻⁹² Ozone is a strong oxidant and widely used for disinfection of water, air, and surfaces, and ozone treatment in hospitals is increasingly used during the ongoing COVID-19 pandemic. Ozone is suggested by molecular modelling to react with specific amino acids and lipid components found in the proteins and lipid bilayer envelope of SARS-CoV-2 virus particles, respectively.²³² In addition to disinfecting surfaces with UV irradiation and chemicals, antiviral material-coated surfaces have been studied for SARS-CoV-2 inactivation.

The inactivation of SARS-CoV-2 on antiviral material-coated surfaces is summarized in Table 5. AgNP-coated non-woven fabric masks and polycotton fabrics have been found to be effective in inhibiting SARS-CoV-2 with a reduction of infectious virus number by approximately 83–99%.^{233,234} An antiviral test of AgNPs with different sizes ranging from 2 to 100 nm showed that the infectious amount of SARS-CoV-2 was reduced by



Fig. 10 Virus inactivation on surfaces coated with 5,10,15,20-tetrakis(1-methylpyridinium-4-yl)porphyrin (TMPyP) as a triplet photosensitizer under visible light irradiation (a). ${}^{1}O_{2}$ is generated from triplet-triplet energy transfer (ET) (b), causing the oxidative degradation of molecular components within virus particles.

Antiviral material	Initial viral titre (N_0)	Virus reduction ratio, $\log_{10}(N_0/N_t)$ (exposure time)	Ref.
AgNPs (2 nm, 2 mg L^{-1})	1.3×10^9 RNA copies	99.2%, 2.1 (1 h)	235
AgNPs (10 nm, 2 mg L^{-1})	1.3×10^9 RNA copies	92.3%, 1.1 (1 h)	235
AgNPs (15 nm, 2 mg L^{-1})	1.3×10^9 RNA copies	90%, 1.0 (1 h)	235
AgNPs (50 nm, 2 mg L^{-1})	1.3×10^9 RNA copies	38.5%, 0.21 (1 h)	235
AgNPs (80 nm, 2 mg L^{-1})	1.3×10^9 RNA copies	53.9%, 0.27 (1 h)	235
AgNPs (100 nm, 2 mg L^{-1})	1.3×10^9 RNA copies	61.5%, 0.41 (1 h)	235
Ag nanocluster (0.55 at%)/silica composite-coated mask	$2.1 imes 10^4 ext{ TCID}_{50} ext{ mL}^{-1}$	91.9%, 1.1 (1.5 h)	234
Ag nanocluster (1.53 at%)/silica composite-coated mask	$2.9 imes 10^4 ext{ TCID}_{50} ext{ mL}^{-1}$	>99.9%, >4 (1.5 h)	234
AgNPs (23.5 nm, 5%)-coated polycotton	1.26×10^9 copies mL ⁻¹	83.1%, 0.77(5 min)	233
Metallic Ag (2 mm) and Zn (1 mm) dot-coated polyester fabric	10^{5} particles mL ⁻¹	60%, 0.40 (1 min)	236
	10^5 particles mL ⁻¹	75%, 0.60 (5 min)	
Clyraguard cuprous iodine complex	$10^{6} \text{ TCID}_{50} \text{ mL}^{-1}$	99.9%, >4 (0.5 h)	250
Annealed Cu (5–60 μm)-coated stainless-steel	$10^{5.1} \text{ TCID}_{50} \text{ mL}^{-1}$	97.9%, 1.8 (5 h)	251
Cu (5–60 µm)-coated stainless-steel	$10^{4.9} \text{ TCID}_{50} \text{ mL}^{-1}$	99.2%, 1.9 (5 h)	251
Cu powder (0.8 \pm 1.0 μ m)	$10^{5.0} \text{ PFU mL}^{-1}$	99.8%, >2.7 (10 min)	239
Luminore CopperTouch [™] (85% Cu)	$1.3 imes10^5~\mathrm{PFU}~\mathrm{mL}^{-1}$	99.8%, 2.8 (2 h)	252
Cu ₂ O/PU-coated glass	$10^{7.8}_{-1}$ TCID ₅₀ mL ⁻¹	99.9%, >3.64 (1 h)	237
Cu ₂ O/PU-coated stainless steel	$10^{7.8}_{-1}$ TCID ₅₀ mL ⁻¹	99.9%, 2.97 (1 h)	237
CuO (30 µm)-coated glass	$10^{7.8}$ TCID ₅₀ mL ⁻¹	99.9%, 3.1 (1 h)	238
Si_3N_4 powder (0.8 ± 1.0 µm)	$10^{5.0}$ TCID ₅₀	>99.9%, >3.0 (10 min)	239
Aluminium nitride powder (1.2 \pm 0.6 μ m)	$10^{5.0}$ TCID ₅₀	>99.9%, >3.7 (10 min)	239
Etched Al alloy with nanostructured surface ($R_{\rm rms}$ = 995 ± 115 nm)	$10^5 \text{ TCID}_{50} \text{ mL}^{-1}$	>99.9%, 5 (6 h)	242
Zn ²⁺ (0.033%)-embedded PA66 fabric	10^4 PFU mL^{-1}	90%, 2 (1 h)	169

90–99% after 1 h exposure to 2 mg L⁻¹ AgNPs with nanoparticle diameters ranging from 2 to 15 nm, while the corresponding reduction levels were 38–62% for AgNPs with diameters ranging from 50 to 100 nm.²³⁵ A luciferase-based pseudovirus entry assay revealed that AgNPs potently inhibited viral entry step *via* disrupting viral integrity. However, cytotoxic effects were observed for AgNPs at concentrations of \geq 20 mg L⁻¹, indicating that improper disposal of Ag-containing products could harm environmental ecosystems. An electroceutical polyester fabric containing Ag and Zn metal dots generated a potential difference of 0.5 V, which resulted in 60–75% inactivation of a model porcine respiratory coronavirus after contact for 1–5 min.²³⁶

Glass and stainless steel coated with a PU film with a thickness of 10–16 μ m that contained 5.1 μ m Cu₂O particles reduced the viral titre by $> 10^{3.64}$ - and $10^{2.97}$ -fold after contact for 1 h, as compared to a reduction of only $10^{0.04}$ -fold for uncoated surfaces. 237 Glass coated with 30 μm CuO reduced SARS-CoV-2 viral titre by 10^{3.1}-fold after contact for 1 h, and the coating remained durable even after exposure to 70% ethanol or 3 wt% bleach.²³⁸ A 15 min exposure to 15 wt% aqueous suspensions of Cu particles, silicon nitride (Si_3N_4) , or aluminium nitride (AlN) showed >99% viral inactivation of SARS-CoV-2 virus particles as well as degradation of viral RNA.²³⁹ The antiviral effect of Si₃N₄ and AlN may be related to the leaching of N atoms from the surface to form gaseous ammonia (NH₃) and ammonium (NH₄⁺) ions through hydrolysis. The similarity between NH4⁺ on Si₃N₄ and the N-terminal of lysine might trigger an effective "competitive binding" mechanism for SARS-CoV-2 virus particle inactivation,²⁴⁰ while eluted NH₃ may penetrate to the inside of virus particles and degrade viral RNA through alkaline trans-esterification via hydrolysis of phosphodiester bonds in the nucleic acid structure.²⁴¹ Notably, the fabrication of Al alloy surfaces with randomly aligned ridges with 23 nm separation

distance reduced the SARS-CoV-2 viral titre by 10⁵-fold after contact for 6 h compared to a flat Al surface,²⁴² highlighting the importance of nanostructure design.

In addition to the antiviral surfaces described above, selfcleaning surfaces coated by other antiviral materials such as QACs, metal ions, visible-light-responsive photocatalysts, and triplet photosensitizers for other enveloped viruses can also be examined for decontamination of SARS-CoV-2 due to the similarity among enveloped virus particle structures.^{243,244} It is noted that envelope proteins, such as the S protein of SARS-CoV-2, are generally amphipathic^{245,246} and negatively charged at physiological pH,^{247,248} and possess several disulphide bonds.^{38,249} For example, the Cys15-Cys136 and Cys840-Cys851 disulphide pairs are found in the SARS-CoV-2 S protein and important for protein folding.⁴⁸ Thus, both positively charged and hydrophobic groups on antiviral surfaces can interact with viral envelope proteins and trigger morphological/conformational changes,^{32,33} while AgNPs, CuNPs, and Au and Cu compounds can in principle inhibit S protein-mediated envelope fusion. Interestingly, the coating of contact surfaces in cars by WO₃ as a visible-lightresponsive photocatalyst is increasing during the COVID-19 pandemic, highlighting the potential for suitable material innovations to be readily deployed once appropriate standards for durability, cost, and scalability are demonstrated among various technical objectives.

6. Conclusions and outlook

Viruses can contaminate surfaces and survive in inanimate environments, although they cannot replicate independently without a host cell. Temperature, humidity, and surface properties influence the duration and degree of virus survival. As temperature

Table 6	Comparison of	f typical s	self-cleaning	surfaces for	decontamination	of viruses ^a
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Self-cleaning surface	Mechanism	Challenges	Future opportunities
SHPB	Repel by "air cushion" between water and SHPB surface ^{94,99}	Incomplete detachment with viral titre reduction of $\leq 10^2$ -fold, low durability due to rubbing	Combine with antiviral materials and guide nanostructure fabrication
QAC-coated	Disrupt PL layer of envelopes by electrostatic and hydrophobic interactions ^{107,108,124-126}	No or less antiviral activity against non-enveloped (NE) viruses with viral titre reduction of $< 10^{0.5}$ -fold for 2 h	Combine with ROS-inducing antiviral materials
Ag ⁺ - or AgNP-coated	Disrupt SH/S–S exchange of envelope proteins by Ag^+ ; ^{109–111} oxidize proteins by ROS generated from reduction of O ₂ by $Ag^{0.148,156,157}$	High cost of Ag, no or less antiviral activity against NE viruses with viral titre reduction of $< 10^{1.5}$ -fold for 1 h	Develop AgNP-coated surfaces with high Ag content and low leaching
Cu ²⁺ -coated	Disrupt SH/S–S exchange of envelope proteins and viral nucleic acid degradation by Cu ^{2+109,113,114,162,163}	No or less antiviral activity against NE viruses with viral titre reduction of $<10^{0.5}$ -fold for 0.5–4 h	Develop composites of CuO and Cu, and/or combine with ROS-mediated antiviral materials
Cu or Cu ⁺ compound-coated	Oxidize proteins by ROS generated from reduction of O_2 by Cu^+ and disrupt SH/S-S exchange of envelope proteins by $Cu^{+119,122,164,166}$	Short-time antiviral activity of Cu^+ as active species released from Cu and Cu^+ compounds at high humidity	Coat surface with 10–50 nm thick polymer films containing firmly attached Cu and Cu ₂ O particles
Photocatalyst-coated	Degrade envelope and capsid proteins during TiO ₂ photocatalysis ^{200,201}	Low inactivating activity under visible light irradiation	Modulate doping, defects, and vacancies of TiO_2 to increase inactivating activity
Visible-light-responsive photocatalyst-coated	Degrade proteins on envelope and capsid during photocatalysis ^{185–188,190,216}	Requires strong light excitation intensity of 100–150 mW cm ⁻² , low inactivating activity	Identify/design photocatalysts with strong absorption and high activity
Visible-light-driven photosensitizer-coated	Degrade proteins on envelope and capsid by ${}^{1}O_{2}$ generated from triplet photosensitizer 193,197,198	Photobleaching of photosensitizer, low inactivating activity	Combine with polymer nanofiber membranes and identify/design photosensitizers with strong absorption and high activity
^a Summarized from Tab	bles 2–4.		

decreases at moderate humidity (40–70%), viruses typically survive for longer periods of time. The exposure of viruses to cold air outdoors and dry air indoors tends to increase virus transmission and spreading, which is consistent with COVID-19 transmission rates observed in various countries worldwide and the generally occurring seasonal flu in winter months.^{253,254} Therefore, the development of self-cleaning surfaces for virus decontamination by inactivation mechanisms is urgently needed as summarized in Table 6.

Although the anti-adhesion property of SHPB surfaces with nanostructures disappears by rubbing, the self-healing and antiviral properties of anti-adhesion surfaces can improve durability and performance reliability. To date, a variety of antiviral surfaces have been prepared to inactivate viruses. For example, QAC-coated surfaces inactivate enveloped viruses but not nonenveloped viruses. In general, Cu-coated surfaces have received more attention than Ag-coated surfaces, due to the lower cost of Cu and related broad-spectrum antiviral activity against both enveloped and non-enveloped viruses. On the other hand, a TiO₂ photocatalyst on surfaces can inactivate viruses and can be used in air filters and membrane-based water filters for virus decontamination under UV light irradiation. Visible-lightresponsive photocatalyst nanofiber membranes can also inactivate viruses under visible light irradiation.

Looking forward, the development of broadly useful antiviral surface coatings that work against multiple viruses would be beneficial to deal with current virus problems and future ones. From this perspective, ROS-mediated oxidation is likely the best inactivation method for developing improved antiviral surface coatings, especially those that utilize Cu and Cu⁺ compounds.^{255,256} Since stainless steel is widely used in hospitals and public facilities, coating steel surfaces with Cu and Cu⁺ compounds can create selfcleaning properties for virus inactivation.^{257,258} While Cu-based coatings are widely used, an important need remains to develop practically useful nanostructured Cu arrays and nanoporous Cu₂O, which can potentially improve viral inactivation performance to a significant extent. In addition to empirical results supporting this notion, the design of nanoporous surfaces also fits with fluid mechanic models.²⁵⁹

Fig. 11 presents illustrative examples of future design directions that could be useful to explore for next-generation antiviral surface coatings. Ultimately, improving performance results will depend



Fig. 11 Possible directions for designing improved antiviral surface coatings. The examples focus on tuning nanostructure geometry and include nanostructured Cu arrays (a) and nanoporous Cu_2O (b) to optimize viral protein-surface contact, thereby increasing the impact of ROS-mediated oxidation. Postfusion S protein structure of SARS-CoV-2 fitted with Protein Data Bank ID of 6XRA and showing densities of N1098, N1074, N1134, N1158, N1173, and N1194 *N*-glycosylation sites (c).³⁷

on various factors such as surface properties, material structure, antiviral materials, preparation method, environment, and virus type and amount. There is still limited mechanistic understanding about how virus-repelling and virus-inactivating surfaces prevent contamination and facilitate decontamination and further investigation is needed, especially if such efforts can be aligned with functional design concepts such as nanoarchitectonics.^{260,261}

While Cu-based coatings are already widely used in real-life settings, most of the advanced material surface coatings described herein have not yet been commercialized. Interdisciplinary research across the fields of virology, chemistry, physics, materials science, surface science, nanotechnology, polymer science, and engineering could further enhance ongoing material innovation in the field and support the development of high-performing, next-generation antiviral surface coatings for practical use and broad application. One promising application scope involves antiviral face masks, including photoactive coatings²⁶² and natural material-based coatings with improved biocompatibility,²⁶³ and can be combined with emerging topics such as stimuli-responsive materials,^{264,265} biomimetic surface functionalization,²⁶⁶ and advanced nanomaterials, *e.g.*, two-dimensional materials.²⁶⁷

Conflicts of interest

There are no conflicts to declare.

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